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## Desulfurization and tar removal from gasifier effluents using mixed rare earth oxides

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DESULFURIZATION AND TAR REMOVAL FROM GASIFIER EFFLUENTS USING  
MIXED RARE EARTH OXIDES

A Thesis

Submitted to the Graduate Faculty of the  
Louisiana State University and  
Agricultural and Mechanical College  
in partial fulfillment of the  
requirements for the degree of  
Master of Science in Chemical Engineering

In

The Department of Chemical Engineering

By  
Sumana Adusumilli  
B.Tech., Andhra University, 2007  
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## ABSTRACT

Biomass gasification is a promising source of fuels. However, hydrogen sulphide, tars and other by-products must be removed from the raw gas because they deactivate downstream reforming and water gas shift catalysts. The goal of this project is to find the best REO combination for simultaneous tar cracking and desulfurization of gasifier effluents and to find the sorbents that are stable at high operating temperatures of gasifiers. Simultaneous tar cracking and H<sub>2</sub>S removal from a simulated gasifier effluent was tested using different rare earth mixed oxide (REO) catalysts/sorbents based on Ce/LaO<sub>x</sub>, Ce/La/MO<sub>x</sub> and Ce/La/M<sub>2</sub>O<sub>x</sub>/Al<sub>2</sub>O<sub>3</sub> where M is a transition metal and M<sub>2</sub> is a third rare earth metal. These catalysts were prepared using sol gel and incipient wetness impregnation methods. Desulfurization tests were done at 903K using a gas composition of 23.4% H<sub>2</sub>, 32% CO<sub>2</sub>, 3.1% H<sub>2</sub>O, 41.4% N<sub>2</sub> and 0.1% H<sub>2</sub>S. The tar cracking/reforming capability of these materials was tested by adding 0.35 mole% naphthalene as a model compound of tar to the simulated effluent and reacting it during the adsorption cycle.

Sorbents containing pure Ce/La oxides have low sulfur capacities and are not very effective in removing H<sub>2</sub>S from a real gasifier effluent. Supporting the REOs on Al<sub>2</sub>O<sub>3</sub> (20 wt% REO) or ZrO<sub>2</sub>, and addition of a small amount of a third REO known to enhance the thermal stability of CeO<sub>2</sub> (either Tb<sub>2</sub>O<sub>3</sub> or Gd<sub>2</sub>O<sub>3</sub>), greatly increased the total sulfur capacities of the REOs. These ternary REOs maintained their capacity over a minimum of four successive runs and were regenerated in air. The tar removal capacity of these sorbents was found to be low in the simultaneous presence of H<sub>2</sub>S, H<sub>2</sub>O and CO<sub>2</sub> and all the sorbents deactivated in 30 mins. A mixed Ce/La/Mn oxide was found to be the best catalyst for simultaneous desulfurization and tar removal.

## CHAPTER 1

### INTRODUCTION AND REVIEW OF LITERATURE

#### 1.1 Biomass Gasification

Biomass gasification involves the partial combustion of biomass to produce gaseous fuels by heating in (typically) air, oxygen, steam, or steam-oxygen mixtures. The product gas contains ash particles, volatile alkali metals and tars as well as synthesis gas. “Tar” is a generic term comprising all organic compounds in the product gas excluding C1-C6 gaseous hydrocarbons (Neeft et al., 2002). Others define tars as any hydrocarbon >C2. The gaseous fuels must be cleaned of tars and particulates. Tars can cause several problems, for example coke formation in the pores of filters, plugging the filters (Aznar et al., 1998).

Biomass feedstocks primarily consist of forest and agricultural residues, urban wood wastes and dedicated energy crops (Torres et al., 2007). The feedstocks are of two general types, cellulosic biomass and proteinaceous biomass. The gas yield after conversion of the protein-containing biomass is often low and severe corrosion has been observed in hydrothermal gasification of protein-containing biomass (Kruse et al., 2005).

Different types of cellulosic biomass have been used to study biomass gasification, including pine wood chips (Aznar et al., 1998), poplar wood (Arauzo et al., 1997), pine sawdust (Garcia et al., 2002), cedar wood (Asadullah et al., 2003), wood sawdust (Waldner et al., 2005), Radiata pine (Tasaka et al., 2007), bagasse (Turn 2007), grass silage (Schmersahl et al., 2007), almond shells (Rapagna et al., 1998), and cattle manure (Schmersahl et al., 2007; Elliott et al., 2004). The biomass is a mixture of different compounds varying in composition. A typical wood contains 3 wt% extractives, 23-35 wt% lignin, 20-22 wt% hemicellulose, and 43-49 wt%



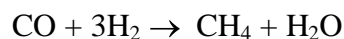
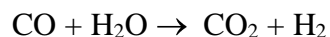
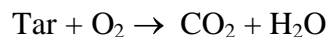
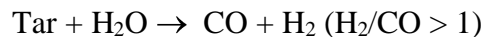
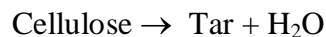
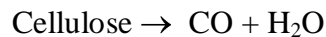
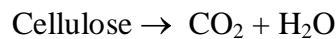
cellulose. Since feeding a real biomass on a laboratory scale is difficult, and because understanding the chemistry of a pure component is easier than understanding that of a mixture, many different model compounds have been used to study gasification (Kruse et al., 2005). Cellulose (Dalai et al., 2003, Asadullah., 2002, Fushimi et al 2003), lignin (Fushimi et al., 2003), and glucose (Kruse et al., 2005) have all been used as model compounds for biomass. The elemental feed composition of biomass for a typical gasification process in wt% is: carbon, 49-52%; hydrogen, 5-7%; nitrogen 0.1-2%; oxygen, 40-43%; sulfur, 0.02-0.3%; chlorine < 0.1% (Pengmei et al., 2007; Zhang et al., 2005; Juutilainen et al., 2006; Arauzo et al., 1994).

The composition of the gas at the exit of a gasifier depends mainly on the type of biomass feed, biomass feeding rate, type of gasifier, gasifying agent (reactant), gasifying agent/biomass ratio, gasifier bed temperature and heating rate (Caballero et al., 1997; Kruse et al., 2005; Asadullah et al., 2002, Fushimi et al., 2003). Because alkali salts catalyze the water-gas shift reaction, the hydrogen yield from biomass gasification is positively affected by a high content of alkali salts in biomass (Kruse et al., 2000). High lignin content adversely affects hydrogen production (Schneider et al., 2000). High gasifier temperatures result in less tar (Kinoshita et al., 1994; Corella et al., 1999). Increasing the gasifier temperature to 1000-1200 K decreases the tar formation by 35% (Ferreira- Aparicio et al., 2005). The temperature range around 1100 K is favorable for gasification of several types of biomass (Milne et al., 2003)

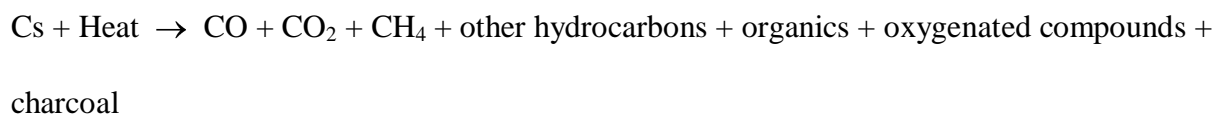
Different gasifying agents (reactants) such as air, steam, steam-oxygen and carbon dioxide have been used. The product gas composition from the gasification of pine wood chips using air at 1053-1113 K with equivalence ratio (ER, the ratio of actual air to fuel ratio to air to fuel ratio required for complete combustion) 0.18-0.45 is 5-16 vol% H<sub>2</sub>, 10-22% CO, 9-19% CO<sub>2</sub>, 2-6% CH<sub>4</sub>, 42-62% N<sub>2</sub>, 11-34% H<sub>2</sub>O, and 0-3% C<sub>2</sub> fraction (Gil et al., 1999). The product

gas composition from the gasification of pine wood chips using steam at 1023-1053 K was 38-56 vol% H<sub>2</sub>, 17-32% CO, 13-17% CO<sub>2</sub>, 7-12% CH<sub>4</sub>, 52-60% H<sub>2</sub>O, and 2% C<sub>2</sub> fraction (Gil et al., 1999). Tar content sharply decreases as ER increases. Higher ER values decrease H<sub>2</sub> and CO concentration in the product gas (Narvaez et al., 1996). At 823 K, only 53% of the carbon was converted in the gasification of cellulose by air. Both CO and H<sub>2</sub> were hardly formed. But by using a Rh/CeO<sub>2</sub> gasification catalyst, 100% carbon conversion was achieved. Ceria itself has catalytic activity in the production of syngas (Asadullah et al., 2001).

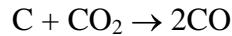
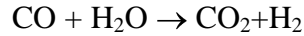
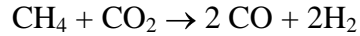
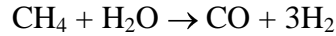
Typical reactions proposed for cellulose gasification by air are (Asadullah et al., 2001):



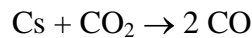
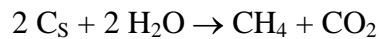
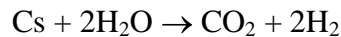
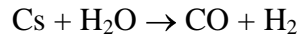
Higher heating rates increase the final conversion of biomass and decrease char production in steam gasification (Fushimiet al., 2003). Increased heating rates significantly increased CO, H<sub>2</sub>, and CH<sub>4</sub> yields in the steam gasification of biomass (Fushimi et al., 2003). Proposed reactions in the steam gasification of biomass are (Raman et al., 1980; Cs represents the carbons in cellulose):



Or if the temperature is sufficiently high, additional reactions take place:



At temperatures above 973K, the following reactions were proposed:



Catalytic steam gasification of almond shells using steam reforming nickel catalysts at 1103 K, GHSV 1800h<sup>-1</sup>, and biomass to steam ratio 1 gives a gas composition of H<sub>2</sub> - 62.1 mol%, CO - 22.7, CO<sub>2</sub> - 15.7%. The gas yield was 1.98 m<sup>3</sup>/kg of biomass and the tar yield was 0.23 g/m<sup>3</sup>; without a catalyst the gas was 1 m<sup>3</sup>/kg of biomass and the tar 100 g/m<sup>3</sup>, at 1043 K (Fushimi et al., 2003). Steam reacts with char above 773 K and does not affect the tar evolution in the low temperature (600-700 K) pyrolysis region. This suggests that pyrolysis is an initial stage of steam gasification (Fushimi et al., 2003).

Pyrolysis at higher temperatures (ca. 773K) can convert biomass to vapors, gases and charcoal in the absence of oxygen. At lower temperatures (673 K), the pyrolysis gas contained mainly CO<sub>2</sub> and CO. Formation of H<sub>2</sub> was less than 2% of CO<sub>2</sub> and formation of CH<sub>4</sub>, C<sub>2</sub>H<sub>4</sub> and

$C_2H_6$  was negligible (Yamaguchi et al., 2006). At 600-700K, 81% of the cellulose converts into tar, also evolving  $CO_2$ , CO and  $H_2$ . (Fushimi et al., 2003).

The product gas composition from the gasification of pine wood chips using both steam and  $O_2$  at 1058-1113 K with ER 0.24-0.51 is 14-32 vol%  $H_2$ , 43-52% CO, 14-36%  $CO_2$ , 6-8%  $CH_4$ , 38-61%  $H_2O$ , 3-4%  $C_2$  fraction (Gil et al., 1999). The gas efficiency of air gasification is 35-70% (Gil et al., 1999), while that of steam - oxygen gasification is about 70% (Caballero et al., 1997).  $CO_2$  gasification in the presence of a Ni/Al catalyst transforms tars, decreases the amounts of  $CH_4$  and  $C_2$  fraction, and increases  $H_2$  and CO yields (Garcia et al., 2000), but the catalyst deactivates rapidly.

Wet biomass contains up to 95% water and results in high drying costs if conventional gasification is used. Hydrothermal gasification is an alternative, either near or above the critical point of water (647 K, 22.1 MPa). The product gas composition from hydrothermal gasification of wood sawdust at 683 K, 29.3 MPa is CO – 9 mol%,  $H_2$  - 16%,  $CH_4$  - 14%,  $CO_2$  - 61%. The carbon gasification efficiency was only 21%. By using a Raney nickel catalyst at 573-683 K and 12-34 MPa, the carbon gasification efficiency was increased to 77-100%. The gas contained 23-48 mol%  $CH_4$ , 43-59%  $CO_2$ , 3-24%  $H_2$  and <0.1% CO. At supercritical conditions, the liquid phase was tar free and contained less than 2% of the feed carbon (Waldner et al., 2005).  $H_2$  and  $CO_2$  formation increased at higher water content;  $CH_4$  formation increased at lower water content (Kruse and Gawlik 2003). An increase in temperature led to an increase in the gasification efficiency. Above the critical temperature of water, a drastic increase in the yields of  $H_2$  and  $CH_4$  and a decrease in CO were found. The pressure dependence of gas formation was only relevant above the critical point of water, where yields of  $H_2$  and  $CO_2$  decreased and the yield of  $C_2$ 's increased with pressure (Kruse and Gawlik 2003).

Biomass, depending on its type, may contain a variety of downstream catalyst poisons, such as sulphur, chlorine and alkaline metals. For example, sewage sludge contains a large amount of sulphur and thus the product gas contains hydrogen sulphide, carbonyl sulphide and sulphur dioxide in high concentrations (Hepola and Simell ., 1997). The product gas composition from gasification of dried sewage sludge in a fluidized bed gasifier containing dolomite and using air and steam at 1123-1173 K is 12-14 vol% H<sub>2</sub>, 7-8% CO, 2-3% CH<sub>4</sub>, 1-2% C<sub>2</sub>H<sub>4</sub>, 16-18% CO<sub>2</sub>, 55-60% N<sub>2</sub>, 18-22% H<sub>2</sub>O, 2-5g/m<sup>3</sup>tar, <100 ppm H<sub>2</sub>S, <200 ppm SO<sub>x</sub>, <10 ppm COS (Sato and Shinoda., 2007).

## **1.2 Biomass Gasifier Catalysts and Their Effects on Product Gas Compositions**

For combustion in most engines, the tar content in fuel gases must be lower than 10 mg/m<sup>3</sup> (Bui et al., 1994). The homogeneous destruction of tars by cracking takes place above 1123 K (Kinoshita et al., 1994). Tar removal from a gasifier product is typically by either or both of two methods: hot gas cleaning after the gasifier (“secondary” method), and treatment inside the gasifier (“primary” method). Increasing the temperature usually decreases the tar content in the product gas. Other operating parameters in the gasifier such as gas reactant, equivalence ratio, residence time etc. also affect the formation and composition of tar. Residual tars can be decomposed by both steam and dry (CO<sub>2</sub>) reforming reactions. Adsorption of tar on the catalyst surface is often the rate determining step of biomass tar decomposition, and catalysts with larger surface areas exhibit higher conversions of tars (Tasaka et al., 2007).

A typical composition of biomass tars in wt% is: toluene - 24%; other one ring aromatic hydrocarbons - 22%; naphthalene - 16%; other two ring aromatic hydrocarbons - 13%; three ring aromatics - 6%; four ring aromatic hydrocarbons-1%; phenolic compounds - 7%; heterocyclic compounds - 10%; others - 2% (Coll et al., 2001). Tasaka et al. (2007) compared tar obtained

from steam gasification of cellulose and steam gasification of real biomass. In steam gasification of cellulose, using a 12% Co/MgO catalyst, all the recovered tar was water soluble. In steam gasification of Radiata pine with the same catalyst, the water soluble tar was only 52%. The different tendencies for tar production and conversion have been attributed to different phases of tar in contact with the catalyst: liquid /solid for cellulose tar, but gas/solid for biomass tar.

Biomass gasifier catalysts (when used) are mainly calcined rocks (e.g., olivine), clay minerals, alkali or alkaline earth oxides, or ferrous metal oxides. All such catalysts affect the gas composition, both decreasing the amount of tars and CO, and increasing H<sub>2</sub> and CO<sub>2</sub> (Caballero et al., 1997). To prevent deactivation of downstream nickel steam reforming catalysts, the tar content of the product gas should be less than 2 g/m<sup>3</sup>. This can sometimes be achieved with dolomite, which decomposes “soft” tars such as phenol derivatives (Cabarello et al., 1997), but not refractive tars such as PAHs, which may actually increase (Narvaez et al., 1997). The most commonly used primary (in the biomass bed) catalytic materials are dolomites, olivines and other calcined minerals (Cabarello et al., 1997; Mastellone and Arena 2008). These all can reduce the amount of tar in the effluent.

Simell et al. (1992) classified calcined minerals according to CaO/MgO ratio. The catalytic activity of such minerals for tar elimination is due to their high alkali (K, Na) content. The activity of these rocks can also be improved by increasing the Ca/Mg ratio, decreasing the grain size, and increasing the content of an active metal such as iron (Simell et al., 1992).

Olivine [(Mg,Fe)<sub>2</sub>SiO<sub>4</sub>] has a higher attrition resistance than dolomite (Rapagna et al., 2000), but its catalytic activity for tar decomposition is lower (Courson et al., 2000). At 1023 K, olivine intercalated with a small amount of Ni<sup>2+</sup> has a high activity for dry reforming (95% methane conversion) and steam reforming (88% methane conversion) (Courson et al., 2000).

Clay minerals belong to the kaolinite, montmorillonite and illite groups (El Rub et al., 2004). The catalytic activity for tar elimination of these clay minerals depends upon the effective pore diameter, surface area and number of acidic sites (Wen and Cain 1984). They typically have lower gasification activity compared to dolomite and cannot be used at temperatures  $>1070$  K (Simell and Bredenberg., 1990).

Metallic iron can also catalyze tar decomposition, more effectively than the oxides. Iron also catalyzes the water-gas shift reaction (Simell et al., 1992); but the activities of magnetite ( $\text{Fe}_3\text{O}_4$ ) and hematite ( $\text{Fe}_2\text{O}_3$ ) catalysts for the decomposition of tarry compounds in fuel gas in the temperature range of 973K – 1173K were lower than that of dolomite. For steam gasification of cellulose using a primary Co catalyst at 873 K, tar conversion increased with Co loading. A 36 wt% Co/MgO catalyst showed 84% tar conversion and 67% carbon conversion to gas. The amount of recovered tar and the elemental composition of the recovered tar was independent of S/C ratio (Tasaka et al., 2007).

### **1.3 Tar Cracking of Gasifier Effluent**

Different model tar compounds have been used to study the “secondary” tar cracking of simulated gasifier effluent. Naphthalene (Furusawa and Tsutsumi 2005; Nacken et al., 2007; Sata and Fujimoto 2007; Bampenrat et al., 2008), toluene (Pansare et al., 2008; Lamacz et al., 2009; Juutilainen et al., 2006), and benzene (Nacken et al., 2007; Furusawa et al 2009) have all been used.

In secondary steam reforming of tar derived from cellulose gasification using 12 wt% Co/MgO (873 K, S/C 0.6-1.3, 0.06 s residence time), 80% of the tars were converted. For steam reforming using a simulated gas containing 3.5 mole% naphthalene, 21 mole%  $\text{H}_2\text{O}$ , 20 mole%

N<sub>2</sub>, and 55.5 mole% Ar, using a 12% Co/MgO catalyst, (1173 K, GHSV 3000h<sup>-1</sup>, S/C ratio 0.6), 23% of the carbon was converted to gas with a final gas composition of 70% H<sub>2</sub>, 27.5% CO<sub>2</sub>, 2.4% CO, 0.1% CH<sub>4</sub>. The catalytic activity of this Co/MgO was greater than that of Ni/MgO (Furusawa and Tsutsumi ., 2005). In later work, Tasaka et al. (2007) tested Co/MgO catalysts with different Co loadings. Catalysts with larger surface area exhibited higher conversion for tar cracking of an actual steam gasifier effluent at 873K. Sato and Fujimoto (2007) reported that a Ni/MgO-CaO catalyst doped with WO<sub>3</sub> as a sulphur-resistant promoter also showed high naphthalene reforming activity, and was stable in gas containing 300 ppm H<sub>2</sub>S at 1073 K, GHSV 14,000h<sup>-1</sup>.

Complete conversion of naphthalene was obtained with Ni/Al<sub>2</sub>O<sub>3</sub> catalysts doped with MgO, using a H<sub>2</sub>S-free synthetic fuel. However, naphthalene conversion decreased below 30% in gasification with a synthetic fuel containing 200 ppm of H<sub>2</sub>S at 1023K and having a face velocity of 2.5 cm/s (Ma et al., 2005). A Ni/Olivine catalyst contacted with 87.75 vol% Ar, 11.6% H<sub>2</sub>O, and 0.7% toluene at 923 K, 3 NL/h gave complete toluene conversion to syngas. However, the product from steam reforming of toluene on just olivine contained polyaromatics (14%), benzene (6%) and methane (2%), formed from CO, CO<sub>2</sub>, and H<sub>2</sub>. The conversion of toluene was only 37% (Swierczynski et al., 2006). Hepola and Simell (1997) also found that the tar cracking activity of Ni-based catalysts decreased as a result of H<sub>2</sub>S adsorption, whereas ammonia conversion was enhanced by a higher H<sub>2</sub>S concentration. High operating temperatures lessened the catalyst deactivation caused by the H<sub>2</sub>S (Hepola and Simell 1997).

Nickel-based tar cracking catalysts are generally placed downstream of the gasifier (Corella et al., 1997). When using typical naphtha steam reforming catalysts, the H<sub>2</sub> and CO



contents increase somewhat, while CH<sub>4</sub> decreases by 0.5 -2.5 vol%, and C<sub>2</sub>H<sub>n</sub> by 1-1.8 vol% (Caballero et al., 2000).

#### **1.4 Catalyst Life**

Long term tests – hundreds of hours - are needed to check the feasibility of the tar cracking catalysts at a commercial scale (Aznar et al., 1998). For commercial steam reforming Ni-based catalysts at 1100 K, there was no deactivation for 45 h on stream in the steam gasification of pine wood chips (Aznar et al., 1998). A MgO-supported Ni catalyst (6% Ni) was active for naphthalene reforming (Nacken et al., 2007). A model gas containing 50% N<sub>2</sub>, 12% CO, 10% H<sub>2</sub>, 11% CO<sub>2</sub>, 5% CH<sub>4</sub>, 12% H<sub>2</sub>O, 0.0875% naphthalene, and 100 ppmv H<sub>2</sub>S was used at GHSV 2080 h<sup>-1</sup> and at 1073K. Complete naphthalene conversion was achieved even after 100 h operation.

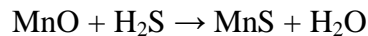
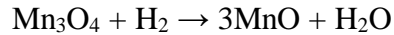
#### **1.5 Mn- and V-Containing Sorbents for Desulfurization**

Removal of sulfur from the gases exiting the gasifier is necessary as it poisons water-gas shift catalysts and poses environmental problems. Mixed oxides based on Zn, Fe and Ti showed good performance for desulfurization, but at high temperatures they were reduced to the metallic state (Desai et al., 1990). Of all the different inorganic oxides tested, MnO exhibited the highest initial reaction rate with H<sub>2</sub>S in the temperature range 573- 1073 K (Westmoreland et al., 1977). MnO is the stable phase prior to sulfidation in this temperature range, and Mn-based sorbents are not reduced to metallic state at high temperatures.

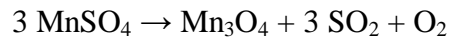
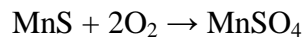
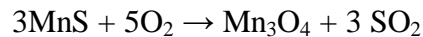
Manganese oxides do not exhibit favourable sulfidation thermodynamics compared to ZnO. But the rate of sulfidation of Mn oxide sorbents was substantially higher than that exhibited

by conventional ZnO-based sorbents (Ben-Slimane and Hepworth 1994). Sulfidation of Mn-based sorbents is high at 1073-1173 K.

The reactions taking place during reduction, sulfidation and regeneration of Mn-based sorbents are (Alonso and Palacios 2002):



Regeneration (using air)

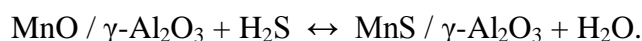


Natural manganese ore consisting mainly of  $\beta$ - $\text{MnO}_2$  is a potential sorbent catalyst for the simultaneous removal of  $\text{SO}_x/\text{NO}_x$ . The main product is  $\beta$ - $\text{MnSO}_4$ . In order to maintain the removal efficiency of  $\text{SO}_2$  and  $\text{NO}$  above 80% and the concentration of  $\text{NH}_3$  in the effluent gas below 5 ppm, the reaction temperature and residence time of the ore was controlled at 623 – 673 K and less than 30 min. The surface area of the sorbent decreased due to formation of  $\text{MnSO}_4$ , which plugged the pores, decreasing the capacity for  $\text{SO}_2$  (Jeong et al., 2001). The addition of  $\text{CuO}$  and  $\text{NiO}$  to the ore resulted in a shorter reduction time and higher sulfidation capacity (Yoon et al., 2003).

Manganese-based sorbents doped with different concentrations of copper showed increased reactivity and stability of the copper oxide, but still thermal sintering (Alonso et al., 1999; Garcia et al., 2000). Sulfidation at 973 K on  $\text{Mn-CuO}_x$  with a gas composed of 0.5 vol%

H<sub>2</sub>S, 10% H<sub>2</sub> and balance N<sub>2</sub> gave a pre-breakthrough H<sub>2</sub>S concentration below 50 ppmv. The presence of Cu in the Mn-based sorbents was necessary to lower the H<sub>2</sub>S concentration to sub ppm levels (Garcia et al., 2000). For MnO doped with ZnO some decay in capacity was observed in 70 sulfidation – regeneration cycles. Sulfidation tests done at 973 K using a gas containing 1 vol% H<sub>2</sub>S, 10% H<sub>2</sub>, 15% H<sub>2</sub>O, 5% CO<sub>2</sub>, 15% CO and balance N<sub>2</sub> gave a pre- breakthrough H<sub>2</sub>S conc. of 10–15 ppmv. Regeneration was with pure air at 1073 K (Alonso and Palacios 2002).

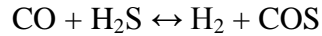
An 8% MnO/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is a regenerable sorbent for the removal of H<sub>2</sub>S (Atakul et al., 1995):



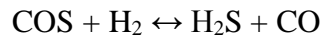
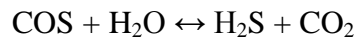
The active compound in this reaction may be MnAl<sub>2</sub>O<sub>4</sub>. The sulfide can be regenerated by steam (Wakker et al 1993). A higher temperature than 873 K increases the breakthrough capacity, but also the deactivation due to sintering. The breakthrough and total capacity of the sorbent were affected by both flow rate and H<sub>2</sub>S concentration, the breakthrough capacity decreasing as the flow rate increased. Mn conversion ranged from 15-19% at breakthrough to 32-35% at maximum sulfidation. The sorbent was completely regenerated at 873 K using N<sub>2</sub>/H<sub>2</sub>/steam mixtures. Thermodynamic calculations show that acceptors with higher manganese content have higher sulfur capacity. Bakker et al., (1996) achieved 17 wt % sulfur capacity with a 32 wt % Mn sorbent. But it could not be easily regenerated. Repeated impregnations of small amounts of Mn gave a high Mn dispersion on alumina, which resulted in high capacity. When the Mn content was increased to 35 wt%, a sulfur capacity of 22 wt % of sulfur was obtained (Liang et al., 1999) at 1123 K for 50% H<sub>2</sub>, 1% H<sub>2</sub>S in Ar. Regeneration was at 1123 K using 30% H<sub>2</sub>O in Ar. No deactivation was observed in 11 sulfidation –regeneration cycles. Another sorbent containing MnO, MnAl<sub>2</sub>O<sub>4</sub> and Mn-Al-O phases showed a sulfur capacity of 20 wt% S, and was stable during 110 sulfidation- regeneration cycles (Bakker et al., 2003). Both MnO and

MnAl<sub>2</sub>O<sub>4</sub> adsorb H<sub>2</sub>S, but MnO adsorbs H<sub>2</sub>S and COS more strongly. MnO adsorbs H<sub>2</sub>S better than MnAl<sub>2</sub>O<sub>4</sub> in the presence of water (Bakker et al., 2003).

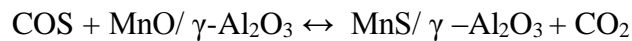
Carbonyl sulfide (COS) is formed by the reaction of CO and H<sub>2</sub>S:



COS is not formed until after H<sub>2</sub>S breakthrough (Wakker et al., 1993), and it increases with high carbon monoxide and low hydrogen concentrations. COS may be removed by direct reaction with the sorbent or it may be converted to H<sub>2</sub>S. In the presence of H<sub>2</sub> and H<sub>2</sub>O, COS is converted to H<sub>2</sub>S as follows:

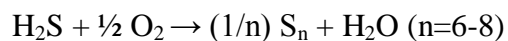


COS reacts with the acceptor as follows:

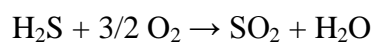


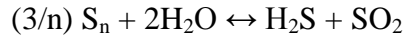
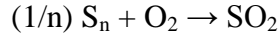
Thermodynamics show that this reaction is favourable at 600-1100 K.

Vanadium-based mixed oxides can oxidize adsorbed H<sub>2</sub>S to sulfur. Among the mixed oxides tested (V/Mo, V/Bi, V/Mg), V/Bi gave the highest sulfur yield. The maximum sulfur yield (97%) was higher than that obtained with vanadium oxide (78%, Li et al., 1996). The reactions taking place in the oxidation of H<sub>2</sub>S are (Terorde et al., 1993):



Side reactions:



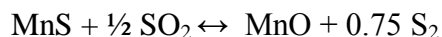
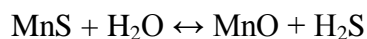


The catalytic performance of the rare earth orthovanadates  $REVO_4$  ( $RE = Ce, Y, La, Sm$ ) was superior to that of  $MgV_2O_6$ , the sulfur yield decreasing in the order  $CeVO_4 > YVO_4 > SmVO_4 > LaVO_4$ . Sulfur yields of both  $REVO_4$  and  $MgV_2O_6$  were much better than those of corresponding single oxides. Temperature programmed reduction showed that the reduction of V cations in  $REVO_4$  was more difficult than in vanadium oxide. XRD measurements indicated the bulk structures of  $REVO_4$  and magnesium vanadates were more stable than vanadium oxide (Li and Chi, 2001).

### 1.6 Regeneration Strategies for Mn-Based Sorbents

Manganese based sorbents can be regenerated using air, steam,  $SO_2$  or an  $SO_2/O_2$  mixture. In contrast, Zn-based sorbents cannot be regenerated using air (Ben-Slimane and Hepworth 1994). The main problem in regenerating Mn-based sorbents is the formation of  $MnSO_4$ , which decreases the capacity of the sorbents in the long run. In oxidative regeneration,  $MnSO_4$  becomes unstable above 1073 K (Ben-Slimane and Hepworth 1994), and is not formed at 1173K. Both the rate of sulfidation and the thermal sintering are not greatly affected by the operating temperature of the sorbent (Garcia et al., 2000).

Steam regeneration can prevent the formation of  $MnSO_4$  and also prevent hot spots. But steam regeneration cannot replace completely the oxidative regeneration process, because it is slow and usually incomplete. Regeneration using steam is also two times slower than  $SO_2$  regeneration (Atakul et al., 1996). The following reactions take place with steam and  $SO_2$ , respectively:



For ZnO-doped MnO, MnSO<sub>4</sub> was formed during the first stages of regeneration, but it decomposed in the last stages of the process (Alonso and Palacios 2002). Sulfided MnAl<sub>2</sub>O<sub>4</sub> can be regenerated using with either SO<sub>2</sub> or H<sub>2</sub>O leading to elemental sulfur or H<sub>2</sub>S. For MnS/Al<sub>2</sub>O<sub>3</sub>, direct regeneration with SO<sub>2</sub> at above 700 K is possible without sulfate formation (Bakker et al., 2003). But regeneration with H<sub>2</sub>O or SO<sub>2</sub> usually requires a lot of regeneration gas.

### **1.7 Rare Earth Oxides (REOs) for Desulfurization and Tar Cracking**

While Ni (on Al<sub>2</sub>O<sub>3</sub>, e.g.) can crack tars to CH<sub>4</sub> and CO<sub>x</sub> at 1100 K, there is rapid coking of the catalyst. However, Ni promoted by CeO<sub>2</sub> shows improved coking resistance (Devi et al., 2003). Similarly, a MgO or basic oxide-doped Al<sub>2</sub>O<sub>3</sub> support also reduces deactivation of Ni-based catalysts by carbon deposition (Bangala et al., 1997).

Among a wide range of metal oxides, CeO<sub>2</sub> was reported to have excellent activity for the oxidation of naphthalene and a high naphthalene adsorption capacity (Garcia et al., 2006). But the problem with CeO<sub>2</sub> as a sulphur adsorbent is its slow adsorption, related to slow redox kinetics (Colon et al., 1998; Flytzani-Stephanopoulos et al., 2006). The reducibility of CeO<sub>2</sub> can be enhanced if intimately mixed with certain other oxides. Substitution of Ce<sup>4+</sup> with Zr<sup>4+</sup> in the CeO<sub>2</sub> lattice improves the oxygen storage capacity, redox properties and thermal resistance. CeO<sub>2</sub> films can be completely reduced at 900 K when supported on YSZ (yttria-stabilized ZrO<sub>2</sub>), and the process is reversible even at higher temperatures (Costa - Nunes et al., 2005). High oxygen mobility and oxygen storage capacity make CeO<sub>2</sub>/ZrO<sub>2</sub> a good catalyst for reforming reactions (Pengpanich et al., 2002; Pengpanich et al. 2006).

CeO<sub>2</sub>-ZrO<sub>2</sub> mixed oxide catalysts showed good activity for the oxidation of naphthalene (Bampenrat et al., 2008). The extent of activity was related to the reducibility. Ce<sub>0.75</sub>Zr<sub>0.25</sub>O<sub>2</sub> showed a high selectivity to CO<sub>2</sub> in the oxidation of naphthalene (Bampenrat et al., 2008). Another benefit of Ce/Zr oxides is their resistance to carbon deposition (Lamacz et al., 2008). A Ni/CeO<sub>2</sub>/ZrO<sub>2</sub> catalyst was also found to be promising for steam reforming of tar (Lamacz et al., 2008). Such catalysts have also exhibited good resistance to carbon deposition and high activity for the steam reforming of methane (Ramirez- Cabrera et al., 2003). A Zr/Al<sub>2</sub>O<sub>3</sub> also shows good activity in the oxidation of tar and ammonia in a biomass gasifier effluent, at low temperatures (below 873K). The presence of H<sub>2</sub>S had little effect on this catalyst (Juutilainen et al., 2006).

### **1.8 Motivation for this Work**

It is hypothesized that mixed REOs (e.g., CeO<sub>2</sub>/La<sub>2</sub>O<sub>3</sub>/Tb<sub>2</sub>O<sub>3</sub>) could simultaneously adsorb H<sub>2</sub>S (to give M<sub>2</sub>O<sub>2</sub>S), crack tars and reform slip methane. The oxides can be regenerated with O<sub>2</sub> at ~900 K (Zeng et al., 1999). For 10-30 at.% La with CeO<sub>2</sub>, the rate constant for the reduction of CeO<sub>2</sub> increases by more than ten times, with reduction substantial at 1070 K (Loong et al., 1999; Bernal et al., 1998). Mixtures of CeO<sub>2</sub> with La are not long-term stable under reducing conditions at >1200 K (Bernal et al., 1998). While CeO<sub>2</sub>/La<sub>2</sub>O<sub>3</sub> is an effective H<sub>2</sub>S sorbent at >873 K, at least initially, it rapidly loses surface area and so sulfur adsorption capacity (>80% after 3 redox cycles, Wang et al., 2005). However, Gd<sub>2</sub>O<sub>3</sub>, Tb<sub>2</sub>O<sub>3</sub> and Sm<sub>2</sub>O<sub>3</sub> dopants similarly increase the rate of reduction (Huang et al., 2005; Bernal et al., 2002), and there are indications that these oxides can thermally stabilize CeO<sub>2</sub>/La<sub>2</sub>O<sub>3</sub>.

Another problem with CeO<sub>2</sub> is the formation of sulfate during oxidative regeneration. While this problem may be alleviated by operating at high space velocities and low sorbent loadings, such conditions are not practical for long-term operation (Flytzani-Stephanopoulos et

al., 2006). Therefore more stable mixed REOs that are active for hot gas cleanup, tar cracking, and as reforming catalysts are needed. It is the purpose of this thesis to explore mixed REOs for these uses. They will be used either as neat mesoporous oxides, or supported on  $\text{Al}_2\text{O}_3$ , or impregnated with an added transition metal oxide ( $\text{MnO}$ ) to further enhance the kinetics of sulfidation.

While it is not necessary to depart entirely from  $\text{CeO}_2$  in order to obtain an oxide mixture active for desulfurization, other REOs are clearly necessary. The mixed  $\text{CeO}_2$  phase should also be more active for tar cracking and further reforming of slip methane than other single REOs such as  $\text{La}_2\text{O}_3$ , because of the excellent redox behaviour of  $\text{CeO}_2$ . Supported (on  $\text{Al}_2\text{O}_3$ ) mixtures of  $\text{CeO}_2/\text{La}_2\text{O}_3$ /third REO may prove superior to the mixed oxides alone, because  $\text{Al}_2\text{O}_3$  can better stabilize (less crystalline ripening) the mixed REO phase when local hot spots occur during regeneration. The goal is to find the best REO combination for simultaneous desulfurization and tar removal from gasifier effluent streams.



## CHAPTER 2

### EXPERIMENTAL

The oxide sorbents/catalysts used here were prepared by either sol-gel (SG) or incipient wetness impregnation methods (IWI).

#### 2.1 Sol –Gel Method

Measured amounts of cerium precursor ceric (IV) ammonium nitrate  $(\text{NH}_4)_2\text{Ce}(\text{NO}_3)_6$  (Aldrich 99.9%; FW = 548.25) and lanthanum precursor La nitrate  $\text{La}(\text{NO}_3)_3 \cdot 6 \text{H}_2\text{O}$  (Alfa Aesar 99.9%; FW = 433.1) salts were added to measured amounts of water and TMAOH surfactant with stirring. The salts dissolved immediately, forming a clear solution. To this solution  $\text{NH}_4(\text{OH})$  (Alfa Aesar, 28-30%  $\text{NH}_3$ ) was added slowly until precipitation occurred (pH ~10.5). The temperature was raised to 363 K and the suspension stirred for 4 days. Every day the pH was checked and brought back to the precipitation pH by adding  $\text{NH}_4(\text{OH})$ . Finally, the precipitate was separated using a centrifuge. The solids were washed with deionised water, acetone, then deionised water, dried at 373 K, then calcined in flowing air at 773 K with a ramp of 2 K/min and a final hold of 6 h. The catalysts prepared in this way are shown in Table 2.1

#### 2.2 Incipient Wetness Impregnation

The precursor salts were either  $\text{Mn}(\text{NO}_3)_2$  (Baker, 50.8 wt% in water),  $(\text{NH}_4)_2\text{Ce}(\text{NO}_3)_6$  (Aldrich, 99.9%; FW = 548.25) or  $\text{La}(\text{NO}_3)_3 \cdot 6 \text{H}_2\text{O}$  (Alfa Aesar, 99.9%; FW = 433.1) were dissolved in deionised water such that the volume of solution in mL was twice the weight of the support oxide in grams. For those supported on  $\text{Al}_2\text{O}_3$  (Engelhard Al-3945E, 1/12") the wt% of alumina in the final oxide mixture was calculated at 80%. The solution was added either dropwise (small batches) or using an orbital shaker (large batches) to the support; the

impregnated oxide was dried at 423 K and then calcined in flowing air at 673 K with a 2 K/min ramp and a final hold of 2 h. The catalysts prepared in this way are shown in Table 2.1.

**Table 2. 1 Sorbent compositions**

Catalyst	Composition(molar)	Method of preparation
REOM_4	Ce/La = 0.9	SG
REOM_14	Ce/La = 3	SG
REOM4_Mn	M/(Ce+La) = 0.1	IWI
REOM4_Mn2	M/(Ce+La) =0.3	IWI
SRE-1	La/Zr = 0.88	IWI
SRE-2	Ce/La/Al = 3/1/53	IWI, 20 wt% Ce/LaO <sub>x</sub> on Al <sub>2</sub> O <sub>3</sub>
SRE-3	Ce/La/Al = 0.9 /1/25	IWI, 20 wt% Ce/LaO <sub>x</sub> on Al <sub>2</sub> O <sub>3</sub>
SRE-5	Tb/Ce/La/Al = 0.2/0.9/1/28	IWI, 20wt% Tb/Ce/LaO <sub>x</sub> on Al <sub>2</sub> O <sub>3</sub>

### 2.3 Characterization of Oxide Sorbents/Catalysts

The BET surface areas of the oxides were measured by N<sub>2</sub> adsorption - desorption using a Quantachrome AS-1 BET apparatus. The oxides were first degassed by heating at 573 K for 1 hour under vacuum. The surface areas were computed from the adsorption branch of the isotherm by a 3-point BET algorithm.

XRD spectra of the powdered samples at high angles were obtained using a Rigaku miniflex 2005C103 X-Ray diffractometer (XRD) using Cu-K $\alpha$  radiation. Samples were scanned from 5-60° at 1°/min with a 0.05° step size. Low-angle XRD spectra were obtained on some samples using the powder XRD beamline at the LSU Center for Advanced Microstructures and Devices using Cu-K $\alpha$  radiation. The samples were scanned from 0.5-10° with a step size of 0.04°, 6 s integration time. The spectra in both analyses were background subtracted and pattern

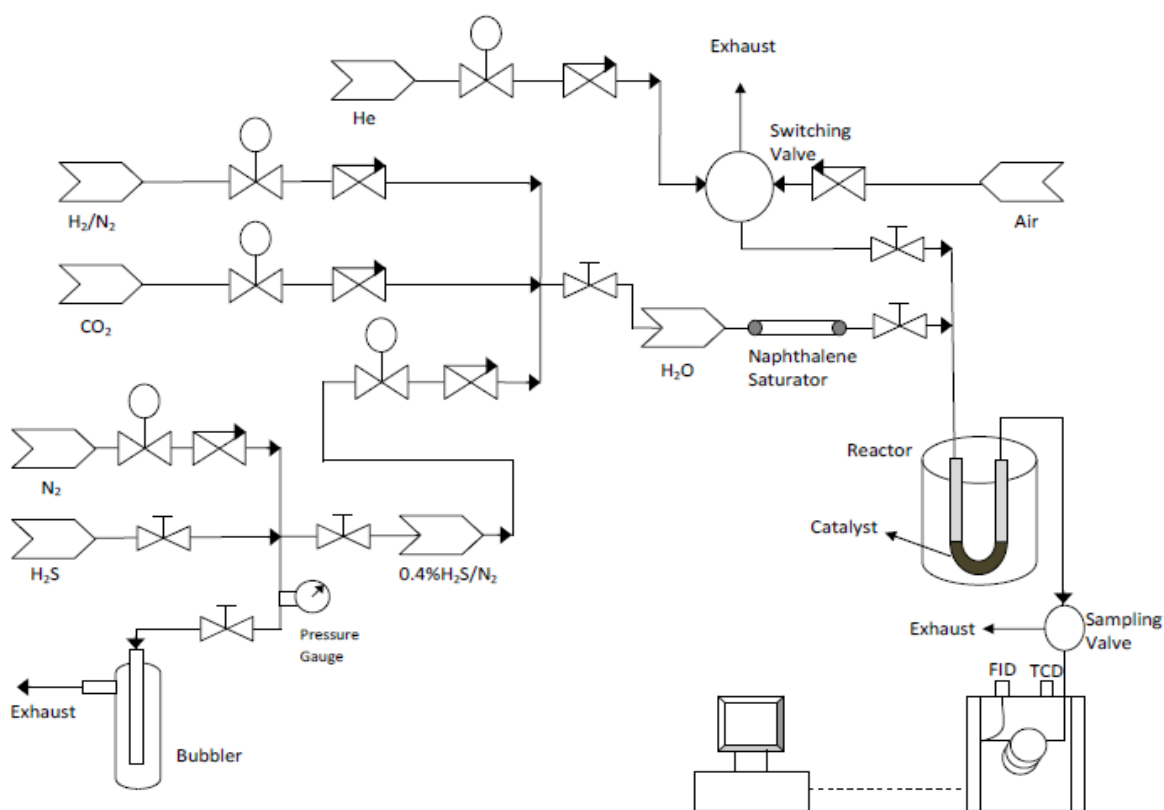
smoothed using MDI Jade software. A peak search algorithm was used to find the exact locations of the peaks.

## 2.4 Tar Cracking Reactions

Microreactor tests for tar (here, naphthalene) cracking were performed using a simulated gasifier effluent of 30.5 mol% CO<sub>2</sub>, 22.2% H<sub>2</sub>, 38.9% N<sub>2</sub>, 8.0% H<sub>2</sub>O, 0.022% H<sub>2</sub>S, and 0.35% C<sub>10</sub>H<sub>8</sub>. To obtain such a low H<sub>2</sub>S concentration a 0.4% H<sub>2</sub>S/N<sub>2</sub> mixture was fed from a lecture bottle. Initially a 2% H<sub>2</sub>S/N<sub>2</sub> mixture was prepared in a lecture bottle from H<sub>2</sub>S (Matheson 99.9%) and N<sub>2</sub> (Airgas, UHP) cylinders. Then the 0.4% H<sub>2</sub>S/N<sub>2</sub> mixture was prepared by mixing appropriate amounts of the 2% H<sub>2</sub>S/N<sub>2</sub> and N<sub>2</sub> cylinders. A 40% H<sub>2</sub> / 60% N<sub>2</sub> mixture (Airgas, grade5) cylinder was used to add H<sub>2</sub>, and a liquified CO<sub>2</sub> cylinder (Airgas, industrial grade) was used to add CO<sub>2</sub>. The flow rates of the gases were adjusted to the required flow rates using manual flow controllers. The gas mixture was then passed through a water saturator maintained at 315 K, and then a naphthalene saturator maintained at 338 K. The naphthalene saturator was a ¼” stainless steel tube with glass wool on both ends. The naphthalene and water saturators were heated using heating tape, and the temperature was controlled using a variable transformer. Two K-type thermocouples were used to measure the temperatures of the saturators. The total gas flow rate was ~110 mL/min at STP. The reacting gas was then passed through the sorbent/catalyst maintained at 903 K. All the catalysts used for the tar cracking reactions had already been used in multiple cycles (regeneration-adsorption- desorption- regeneration) of desulfurization using essentially the same feed, minus the naphthalene.

The catalyst/sorbent itself (0.6-1.0 g with 0.001 g precision) was contained in a ¼” stainless steel U-tube filled with quartz wool on both ends. The U-tube was heated using a sand-filled furnace, controlled using a Eurotherm 818-p PID controller. A K-type thermocouple

measured the temperature of the sand bath. Prior to reaction the oxides were heated to 903 K in air (Industrial grade) flowing at 60 mL/min for 40 min and then the flow switched to He by a Valco 8-port valve operated by Red Lion Libra Timer. The He flow was turned off after 5 min and the reacting gas flow was started. The transfer lines were all maintained at 403 K to prevent condensation of naphthalene and water vapor. The gas was sampled using a 6-port Valco valve operated by the same timer. The gas was exhausted to a fume hood. A schematic of the system is shown in Figure 2.1.



**Figure 2. 1 Schematic of reactor system for tar cracking reactions**

The gas from the sampling valve was analyzed in a HP 5800 Series-II GC. Samples were taken every 2 minutes, until the naphthalene concentration in the exiting gas was equal to that of the inlet gas. Further details on the GC analysis are given in Appendix A.

## 2.5 Sulfidation Tests

Sulfidation (adsorption) tests were performed at 903 K using a gas composition of 23.4 mol% H<sub>2</sub>, 41.4% N<sub>2</sub>, 3.1% H<sub>2</sub>O, 32.0% CO<sub>2</sub>, and 0.1% H<sub>2</sub>S and GHSV of 15500h<sup>-1</sup>. The gas mixture was prepared in the same way as described in the tar cracking reaction tests. For adding H<sub>2</sub>S, a 2% H<sub>2</sub>S/N<sub>2</sub> lecture bottle cylinder was used. The total flow rate of the gas was 100 mL/min at STP. The gas mixture from the cylinders was passed through the water saturator maintained at 298 K, then to the ½” stainless steel U-tube containing about 0.6-1.0g (with 10<sup>-8</sup> precision) of oxide with quartz wool on both ends. The U- tube was controlled as in the tar cracking reactions. The exit gas was passed to a 10-port Valco sampling valve and analyzed using a sulphur-specific pulsed flame photometric detector (PFPD) attached to a Varian 3800 GC. Further details on the analysis are given in Appendix A. The sampling valve was maintained 373 K, and samples were taken until the sulfur concentration in the inlet gas was equal to that in the exit gas for at least 5 min. The exit gas from the sampling valve was exhausted to a fume hood. Prior to the tests, the catalysts were pretreated in the same way as in the tar cracking tests.

## 2.6 Temperature Programmed Desorption and Regeneration

After saturation of the sorbent, the feed flow was switched to He flowing at 60 mL/min. The temperature of the U-tube was raised from 703 to 1103 K at 10 K/min. The gas was sampled every 15 s. The sorbent was held at 1103 K until no sulfur was detected in the exiting gas, and it was then cooled to room temperature in He. All oxides used for either desulfurization or tar cracking experiments were regenerated in air flowing at 60 mL/min, 903K, for 40 min.

## CHAPTER 3

### RESULTS AND DISCUSSION

#### 3.1 Characterization of Materials

Surface areas were measured for both fresh sorbents and for sorbents after use in multiple cycles of sulfidation/regeneration. These are shown in Table 3.1.

**Table 3. 1 Surface area of sorbents before and after used in multiple cycles of sulfidation**

Sorbent	Composition	Surface area after calcination (m <sup>2</sup> /g)	Surface area after multiple sulfidation/TPD tests (m <sup>2</sup> /g)
REOM_14	Ce/La=3	242	98
REOM_4	Ce/La=0.9	110	40
SRE-1	La/Zr=0.8	65	43
SRE-2	Ce/La/Al=3/1/53	160	NA
SRE-3	Ce/La/Al=0.9/1/25	160	150
SRE-4	Gd/Ce/La/Al=0.2/0.9/1/28	160	120
SRE-5	Tb/Ce/La/Al=0.2/0.9/1/28	170	99
REOM4_Mn2	M/Ce+La=0.3	62	5

The average pore diameter calculated for REOM\_14 using the desorption curve and the Barrett-Joyner-Halinda algorithm (BJH) is 3.83 nm and the pore volume is 0.23 cc/g (Kalakota 2008). The surface area of the unsupported REOs (REOM\_4, REOM\_14, REOM4\_Mn2) decreased significantly during the sulfidation runs. This is due to the sintering of the REO nanoparticles or crystallites taking place at the high operating temperatures and high water

partial pressures used in the sulfidation runs. REOM\_14 (Ce/La=3) retained more surface area than REOM\_4 (Ce/La=0.9) indicating that sorbents containing high Ce/La are more resistant to sintering during the sulfidation runs. The REOs supported on Al<sub>2</sub>O<sub>3</sub> and ZrO<sub>2</sub> (SRE-1, SRE-3, SRE-4, SRE-5) showed less reduction in surface area, since both supports increase the thermal stability of REOs (Yi et al., 2005; Trovarelli et al., 1997). Addition of third REO did not improve the surface area retention of the sorbents as was shown by SRE-4 and SRE-5. Reom4\_Mn2 showed a very high percentage decrease in surface area, so possibly the surface Mn aids the sintering process.

The presence of crystalline phases and the average particle sizes of the crystallites were determined using XRD. The average particle size was determined using the Scherrer equation.

$$\beta = K \lambda / L_w \cos(\theta)$$

Where

$$K = 0.94$$

$\lambda$  = wavelength of CuK<sub>α</sub> radiation

$L_w$  = the full width of the peak at half maximum (FWHM)

$\theta$  = the Bragg angle

The distance between atomic layers in a crystal (d) is calculated using Bragg's law:

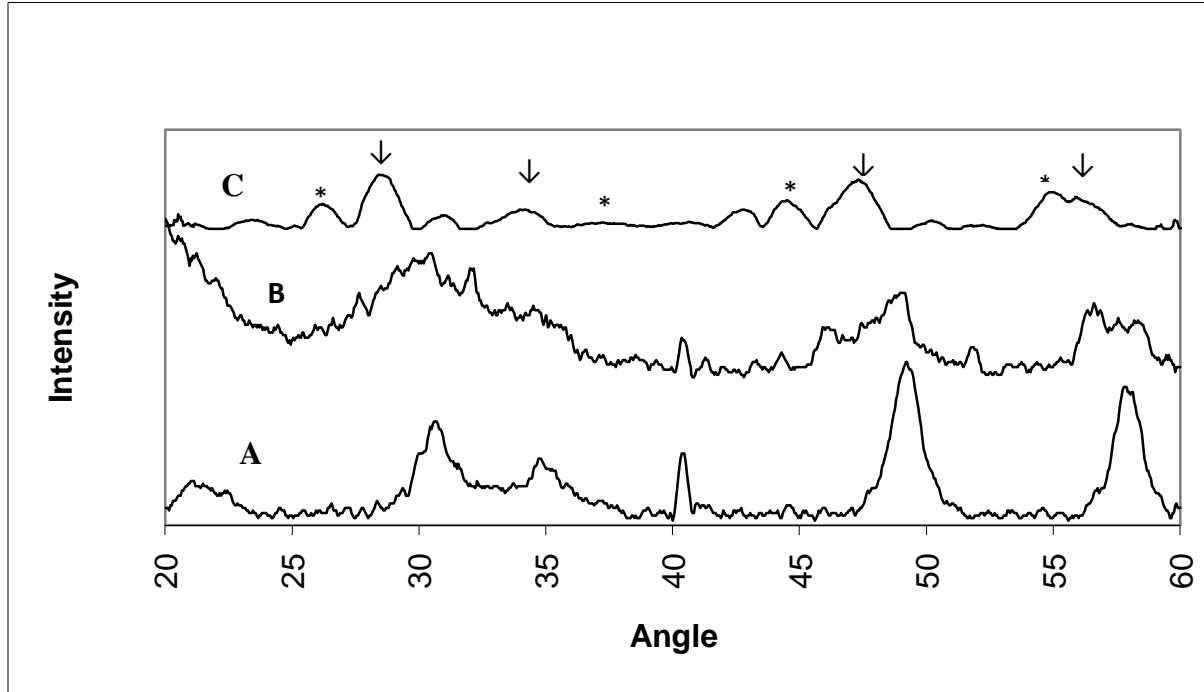
$$n \lambda = 2 d \sin\theta$$

Where

n is an integer

$\lambda$  = wavelength of  $\text{CuK}_\alpha$  radiation

$\theta$  = the Bragg angle



**Figure 3. 1 XRD analysis of Mn-containing sorbents (as calcined): (A) REOM\_4 (B) REOM4\_Mn (C) REOM4\_Mn2**

\*indicates  $\beta\text{-MnO}_2$  peaks

↓ indicates  $\text{CeO}_2$  peaks

REOM\_4 ( $\text{Ce/La}=0.9$ ) was scanned from  $20\text{-}60^\circ$  with  $0.05^\circ$  step. The peaks at  $30.7^\circ$ ,  $34.9^\circ$ ,  $49.3^\circ$ ,  $58.1^\circ$  correspond to (1 1 1), (2 0 0), (2 2 0) and (3 1 1) reflections of cubic fluorite  $\text{CeO}_2$  phase (JCPDS 43-1002). The peak at  $40.5^\circ$  which is visible in all  $\text{Ce/LaO}_x$  could not be identified. The particle size calculated using (1 1 1) reflection of  $\text{CeO}_2$  phase is 8.7nm. The sorbents containing La are shifted towards higher  $2\theta$  values by  $\sim 2^\circ$  compared to pure  $\text{CeO}_2$  (Kalakota 2008). The shifts towards higher  $2\theta$  indicate formation of a ceria-lanthana solution (Bernal et al., 1998).

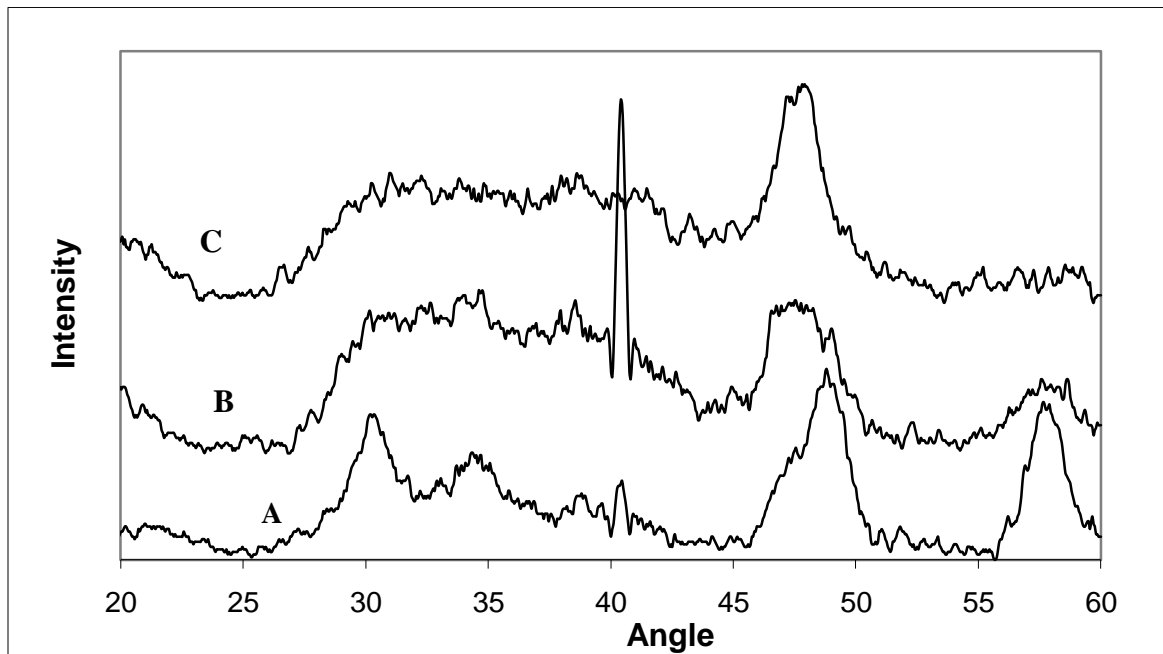


REOM4\_Mn (M/(Ce+La)=0.1) was also scanned from 20-60° degrees with 0.05° step. The peaks are broad denoting the lack of a significant “long-range” crystalline order. The peaks at 30.5°, 34.7°, 49.2° and 57.9° correspond to (1 1 1), (2 0 0), (2 2 0) and (3 1 1) reflections of the cubic fluorite CeO<sub>2</sub> phase. Like REOM\_4, the peaks are shifted towards higher 2θ values by 2°. The presence of a separate LaO<sub>x</sub> and MnO<sub>x</sub> like phase was not detected. This indicates that both La and Mn have been in large part dispersed into the ceria phase. The particle size calculated using the (1 1 1) reflection of the CeO<sub>2</sub> phase is 2.5nm.

REOM4\_Mn2 (M/(Ce+La)= 0.3) was scanned from 20-60° with 0.04° step and 4 s integration time. The phases identified in REOM4\_Mn2 are the cubic fluorite CeO<sub>2</sub> and β-MnO<sub>2</sub>. The peaks at 28.6°, 34.3°, 47.3° and 56° correspond to (1 1 1), (2 0 0), (2 2 0) and (3 1 1) reflections of the CeO<sub>2</sub> phase respectively (JCPDS 43-1002). The peaks at 26.3°, 37.3°, 44.6° and 55° correspond to the (1 1 0), (1 0 1), (1 1 1) and (2 1 1) reflections of the β-MnO<sub>2</sub> phase (JCPDS 24-0735). The peaks shifted towards lower 2θ values by 0.5°. The particle size calculated using the (1 1 1) reflection of CeO<sub>2</sub> phase is 8.93 nm. The particle size calculated using (1 1 0) reflection of the MnO<sub>2</sub> phase is 10.6 nm. The large particle sizes are consistent with the low surface area of the “as calcined” material in Table 3.1.

The alumina-supported REOs, SRE-2 (Ce/La/Al= 3/1/53), SRE-3 (Ce/La/Al= 0.9/1/25) and SRE-5 (Tb/Ce/La/Al= 0.2/0.9/1/28), were scanned from 20-60° with a 0.05° step. Most of the peaks are broad denoting the lack of a significant “long-range” crystalline order. The phase identified in SRE-2 is CeO<sub>2</sub>. The (1 1 1), (2 0 0), (2 2 0) and (3 1 1) reflections of CeO<sub>2</sub> phase were observed at 30.45°, 34.7°, 49.05° and 58.05°. The peaks were shifted towards higher 2θ values by 2°. This is consistent with the formation of a ceria- lanthana solid solution (Bernal et al., 1998). The particle size of SRE-2 calculated using the (1 1 1) reflection of the CeO<sub>2</sub> phase is 6.6 nm. The broad peak between 46° and 51° observed in SRE-2,3,5 is likely an overlap of the (2

2 0) reflection of CeO<sub>2</sub> phase and a reflection characteristic of the  $\gamma$ - Al<sub>2</sub>O<sub>3</sub> phase (JCPDS 29-63). SRE-3 and SRE-5 showed mostly low intensity and broad peaks of the CeO<sub>2</sub> phase indicating low crystallinity. The high intensity peak at 40.45° in SRE-3 could not be identified.



**Figure 3. 2 XRD analysis of supported Ce/La sorbents: (A) SRE-2 (B) SRE-3 (C) SRE-5**

SRE-1 was also scanned from 20-60° with a 0.05° step. It contained a mixture of the monoclinic, tetragonal and cubic crystalline structures of zirconia (Juutilainen et al., 2006). The (1 1 1) reflection of t-ZrO<sub>2</sub> was observed at  $2\theta = 32.35^\circ$ . The reported position for the (1 1 1) reflection of t-ZrO<sub>2</sub> is at  $2\theta = 30.306^\circ$  (Barshilia et al., 2008). The peaks at 30.45°, 33.55° and 57.43° correspond to (1 1 1), (0 0 2) and (0 1 3) reflections of m-ZrO<sub>2</sub> (JCPDS 13-307). The peaks at 36.9° and 52.3° correspond to peaks of the cubic ZrO<sub>2</sub> phase (JCPDS 27-997). No LaO<sub>x</sub> diffraction peaks were observed. This indicates that lanthana formed a solid solution with zirconia such that the only XRD detectable phase is ZrO<sub>2</sub>. The particle size calculated using the (1 1 1) reflection of t-ZrO<sub>2</sub> is 19.1 nm.

The supposedly mesoporous REOs REOM\_4 (Ce/La= 0.9) and REOM\_14 (Ce/La= 3) were scanned from 0.5-10° with a 0.02° step in order to estimate the dominant pore size from the largest d-spacing. NIST Mica 675 standard was used to verify the correct 2θ offset. The large decay at 0.5-0.85° is characteristic of the instrument and not associated with the sample. For REOM\_4, the d-spacing calculated using Bragg's law for the peak at 2θ = 0.92° is 9.6 nm. For REOM\_14, the d-spacing calculated for the peak at 2θ = 1.24° is 7.1 nm. The average pore diameter for REOM\_14 is 3.83 nm. Therefore the calculated total wall thickness for REOM\_14 is 3.3 nm. The total wall thicknesses of CeO<sub>2</sub> mesopores from the literature are reported to be between 3 and 5 nm (Chane-Ching et al., 2005).

### **3.2 Sulfur Adsorption and TPD Tests**

The sulfur adsorption capacity of the sorbents was determined using a synthetic gasifier effluent reaction mixture containing 23.4 mol% H<sub>2</sub>, 41.4% N<sub>2</sub>, 3.1% H<sub>2</sub>O, 32.0% CO<sub>2</sub>, and 0.1% H<sub>2</sub>S. Adsorption tests were done at 903K and atmospheric pressure. TPD tests were done using He as carrier gas from 903K to 1103K. The former tests gave the maximum sulfur adsorption capabilities of the sorbents while the latter tests gave the amount of sulfur that can be easily desorbed. Since both the stainless steel U- tube and the quartz wool adsorb sulfur, blank tests were done first in order to find the exact amount of sulfur adsorbed by these. The amount of H<sub>2</sub>S adsorbed only by the sorbent was then determined by subtracting the amount obtained from the blank tests at the respective times. In order to eliminate fluctuations in the runs, multiple blank tests were performed and the average of 6 runs was used. Blank runs were performed by passing the reaction mixture through the U-tube for 5 minutes and then switching to He and ramping the temperature from 903K to 1103K.

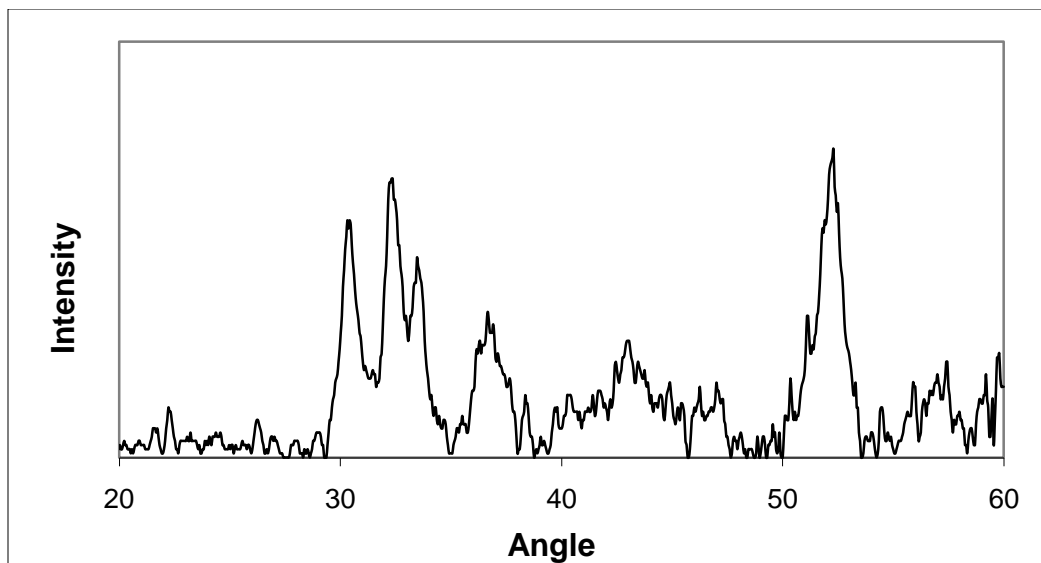


Figure 3. 3 XRD analysis of SRE-1

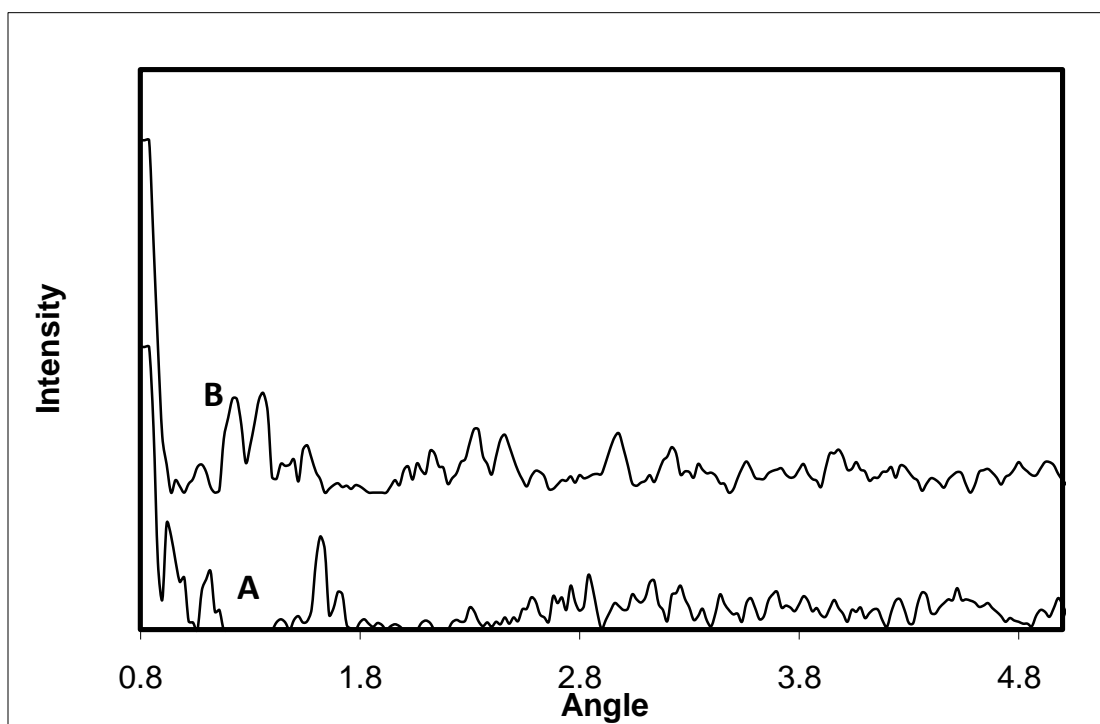


Figure 3. 4 XRD analysis of sorbents. (A) REOM\_4 (B) REOM\_14

$$A_B = (A_1 + A_2 + A_3 + A_4 + A_5 + A_6) / 6$$

Where  $A_1, A_2, A_3, A_4, A_5, A_6$  are the areas (proportional to the amount of sulphur) given by the GC detector at time  $t$  during the six blank tests.

For the non-blank runs:

$$A_S = A_A - A_B$$

Where,

$A_A$  is the area given by the GC at time  $t$  of the actual run using sorbent

$A_S$  is the corrected area for the absolute amount of sulfur.

The corrected areas were then smoothed using a three point average over 0.5 min:

$$A_T = (A_{s-0.25} + 2A_s + A_{s+0.25}) / 4$$

Where,

$A_{s-0.25}$  is the area given by GC at time  $t-0.25$

$A_s$  is the area given by GC at time  $t$

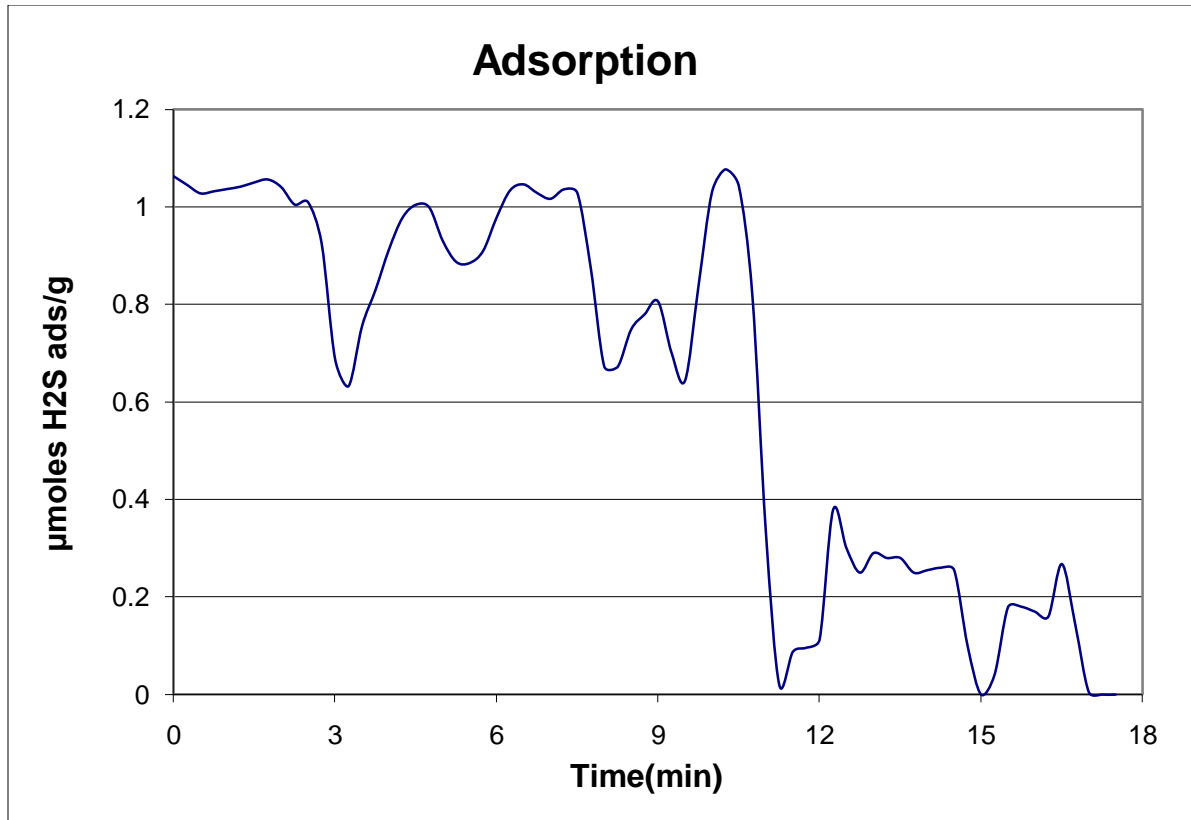
$A_{s+0.25}$  is the area given by GC at time  $t+0.25$

$A_T$  is the corrected area at time  $t$

The micromoles of sulfur at time  $t$  ( $\mu_t$ ) were then found from:

$$\mu_t = A_T \text{ CF}$$

Where CF ( $\mu\text{mole}/\text{Area}$ ) is the calibration factor - see Appendix A for how this was obtained.



**Figure 3. 5 Amount of H2S adsorbed vs time for REOM4\_Mn (4th run).**

The micromoles of sulfur/g of sorbent exiting the reactor from t-1 to t was calculated as:

$$\mu_t = [(\mu_{t-1} + \mu_t) / 2] (F_G / V_s) (t_s / W_s) (T_s / T_0)$$

Where

$F_G$  is the total gas flow rate at STP

$V_s$  is the volume of sampling loop

$t_s$  is the time increment between samples

$W_s$  is the weight of sorbent

$T_S$  is the temperature of sampling loop, 373K

$T_0$  is 273 K

$\mu_T$  is the micromoles of sulfur/g of sorbent exiting the reactor at time  $t$

The total micromoles of sulfur desorbing from the sorbent is found from:

$$\mu_{\text{total}} = \sum_{t=0}^{t_f} \mu_t$$

where  $t_f$  is the time of the desorption of removable sulfur

Total micromoles of sulfur entering the reactor during a run ( $\mu_e$ ) is found as:

$$\mu_E = (\text{Flow rate of H}_2\text{S}) (1/22400) (T_0/T_A) t$$

where

$t$  is the time to attain saturation of the sorbent

$T_A$  is ambient temperature, 298K

The total micromoles of sulfur adsorbed is then:

$$\mu_A = \mu_E - \mu_{\text{total}}$$

Several sorbents were tested in multiple cycles of adsorption- desorption- regeneration. The sulphur adsorption capacities of REOM\_4, REOM\_14, SRE-5, SRE-2, SRE-3 and REOM4\_Mn are shown in Figures 3.6, 3.7 and 3.8. For each sorbent, the cycle was repeated until the adsorption capacity showed increases of less than 30%. The maximum capacities for SRE-1 and REOM4\_Mn2 were previously measured by Kalakota (2008). These maximum sulfur capacities were compared to a commercial BASF Selexsorb CDX 7 X 14 mesh sorbent

composed of Al<sub>2</sub>O<sub>3</sub>/Zeolite with 15 – 40% Zeolite of unspecified phase. The adsorption and desorption capacities of this sorbent (Kalakota 2008) are shown in Figure 3.6. The initial capacity of this sorbent is very high but the capacity decreased sharply in the following run; it is not stable at these conditions.

The sulfur capacities of the supported rare earth oxides were in general found to be greater than that of the unsupported REOs, especially if compared on a basis of per weight of REO - the active weight of the SREs is only 20% of the total sorbent weight. This is because supporting monolayers or nanoparticles of REOs on either Al<sub>2</sub>O<sub>3</sub> and ZrO<sub>2</sub> supports improves the thermal and steam stability of the REOs (Yi et al., 2005; Trovarelli et al., 1997). However, the Al<sub>2</sub>O<sub>3</sub> on which SRE2-5 were supported has very low sulfur capacity.

Among the sorbents tested, REOM4\_Mn (M/(Ce+La)= 0.1) gave the highest and REOM\_14(Ce/La= 3) gave the lowest sulfur capacities. Impregnation of REOs with Mn greatly improved the sulfur removal capacity of the sorbents. The capacity of this sorbent is in the range 90-180 μmoles of H<sub>2</sub>S /g of sorbent; sulfur capacities of REOs impregnated with transition metals are in the order Mn > Fe >> Cu (Kalakota 2008). The capacities of REOs tested here are in the range 20-50 μmoles of H<sub>2</sub>S/g of sorbent. The observed capacities of Ce and La mixed sorbents are in the range of 25 to 250 μmoles of H<sub>2</sub>S adsorbed/g of sorbent for a gas composition of 0.1% H<sub>2</sub>S, 50% H<sub>2</sub>, 10% H<sub>2</sub>O, balance He at 923 K (Flytzani-Stephanopoulos et al., 2006; Wang and Flytzani-Stephanopoulos, 2005). From this it can be concluded that the unsupported Ce/La REOs are not very effective in removing H<sub>2</sub>S from a more realistic gasifier effluent containing CO<sub>2</sub>. At monolayer loading the adsorption capacity of 25wt% Mn/Al<sub>2</sub>O<sub>3</sub> using the same feed used in this work and tested at 873K is 29μmole of H<sub>2</sub>S/g of sorbent (Kalakota 2008). For Ce/Mn oxide sorbents with a Ce/Mn ratio varying from 0.66 to 3, for a gas



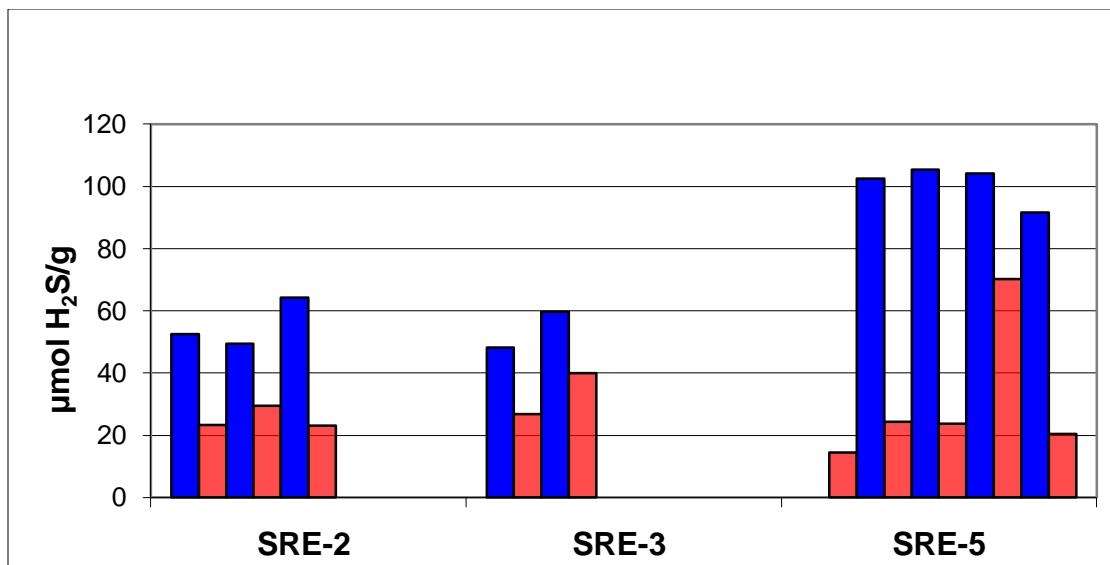
composition of 1% H<sub>2</sub>S and 10% H<sub>2</sub>, balance He the observed capacities for multiple runs are in the range of 610 to 1150 μmole S/g of sorbent at 873 K (Yasyerli, 2008). The capacities of the Mn sorbents tested in this work were lower probably because the feed gas also consists of CO<sub>2</sub> and H<sub>2</sub>O, both of which compete for active sites on the sorbent and decrease capacity.

Among the supported rare earth oxides (SREs), SRE-5 (Tb/Ce/La/Al= 0.2/0.9/1/28) containing a small amount of Tb<sub>2</sub>O<sub>3</sub> gave the highest sulfur removal capacity. This indicates that addition of a third REO improves the sulfur removal capacity. This may be due to the increased oxygen vacancies provided by the third REO (Bernal et al., 2002; Huang et al., 2005). For SRE-5, the sorbent capacity increased in the first three runs and then stabilised. This may be due to the formation of solid solution of the three REOs present in this sorbent. The capacity of this sorbent is slightly greater than a similarly prepared Gd/Ce/La/Al<sub>2</sub>O<sub>3</sub> sorbent (Kalakota 2008).

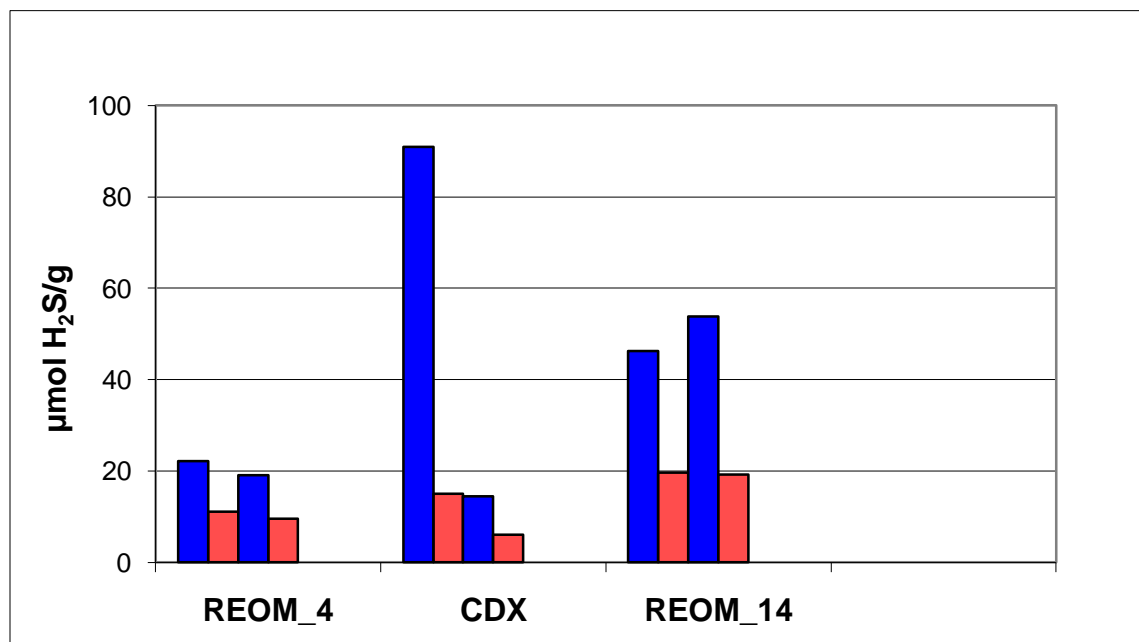
From the TPDs, the amounts desorbed in inert gas were calculated and these are shown in red in Figures 3.6 - 3.8. The amount of sulfur desorbed was approximately 50% of that adsorbed. From the TPDs it is clear that inert gas itself is not sufficient to remove all the adsorbed sulphur, but that much of it is weakly bound. This is in agreement with the literature for REOs (Wang and Flytzani Stephanopolous, 2005).

### **3.3 Tar Cracking / Removal**

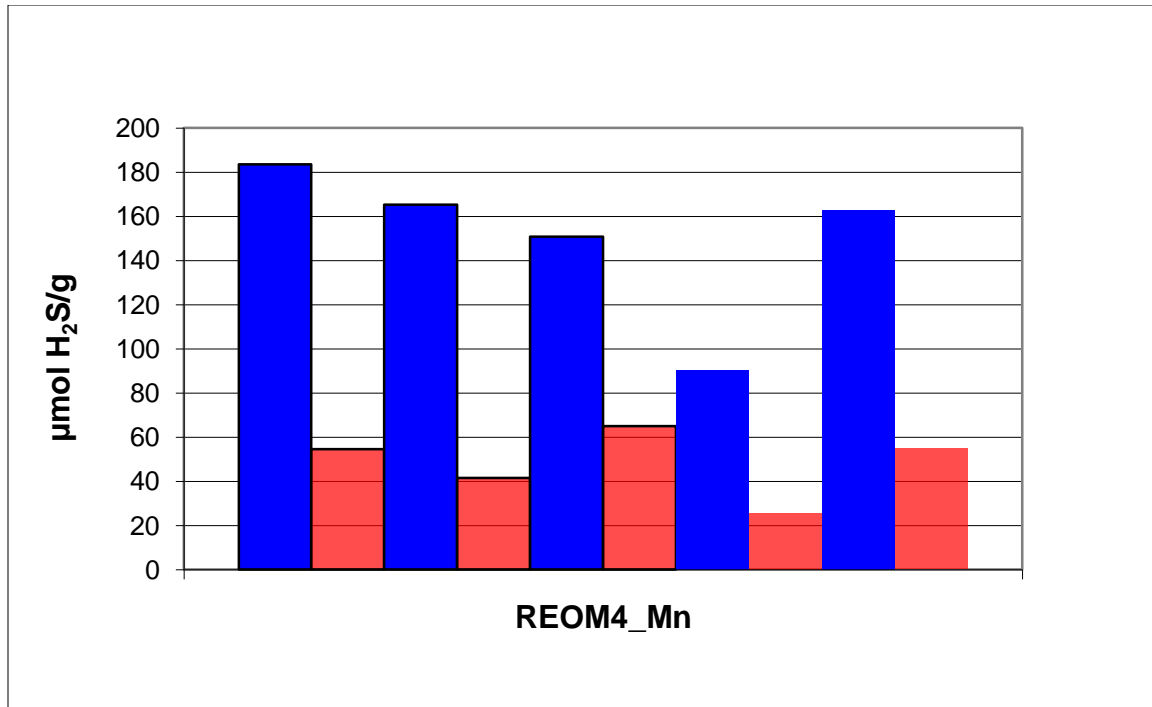
Simultaneous tar cracking and desulfurization experiments were performed using a gas composition similar to that of the previous experiments: 30.5 mol% CO<sub>2</sub>, 22.2% H<sub>2</sub>, 38.9% N<sub>2</sub>, 8.0% H<sub>2</sub>O, 0.022% H<sub>2</sub>S, and 0.35% C<sub>10</sub>H<sub>8</sub>. To measure the tar cracking capability of the non-sulfided catalysts, experiments were run using essentially the same molar composition, but with N<sub>2</sub> replacing H<sub>2</sub>S. All of the sorbents used in the tar cracking tests had already been used in



**Figure 3. 6 Adsorption (dark) and desorption (light) capacities of SRE-2, SRE-3 and SRE-5 sorbents.**



**Figure 3. 7 Adsorption (dark) and desorption (light) capacities of Reom\_4, CDX, Reom\_14 sorbents.**



**Figure 3. 8 Adsorption (dark) and desorption (light) capacities of REOM4\_Mn.**

multiple sulfidation cycles, except where noted in Tables 3.1-3.2 below. All of the tar cracking experiments were done at 903 K and atmospheric pressure.

The % of naphthalene adsorbed and reacted at time t was calculated as:

$$R_t = 100 - (A_t / A_A) (100)$$

Where

$A_t$  is area measured by GC at time t

$A_A$  is the average of the areas of the naphthalene feed

The moles of naphthalene removed (reacted and/or adsorbed) at time increment n ( $M_n$ ) was calculated as:

$$M_n = (R_t/100) (CF A_A / M_B) y_f (F_G / 22400) (273/298)$$

Where,

CF is the calibration factor,  $\mu\text{moles naphthalene/area}$ .

$M_B$  is the maximum number of moles of naphthalene in the sampling loop (for the feed)

$y_f$  is the mole fraction of naphthalene in the feed

$F_G$  is the flow rate of reaction gas mixture

The amount of naphthalene reacted over the time period  $\Delta t$  was then calculated by the trapezoidal rule.

$$\mu_t = [\Delta t] [(M_n + M_{n-1}) / 2] 10^{-6}$$

The total micromoles ( $\mu_{\text{total}}$ ) of naphthalene reacted over 30 minutes  $t_f$  was calculated as:

$$\mu_{\text{total}} = \sum_{t=0}^{t_f} \mu_t$$

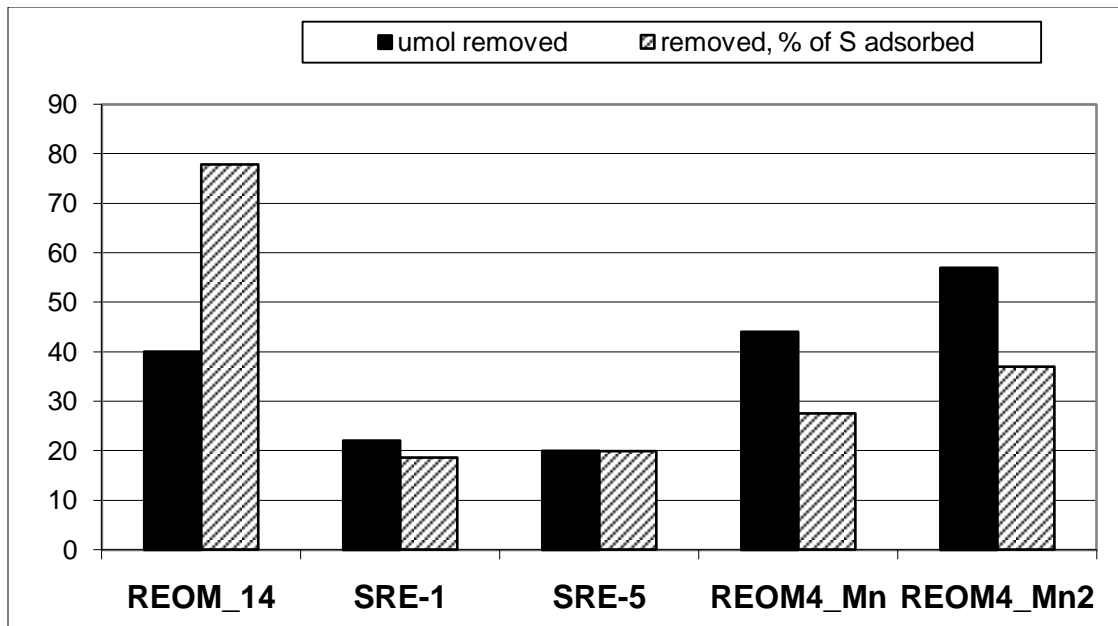
**Table 3. 2 Tar removal of naphthalene and sulfur capacities of sorbents used for multiple sulfidation cycles**

Sorbent	Composition	$\mu\text{moles of tar removed in 30 min with feed containing H}_2\text{S}$	$\mu\text{moles removed in 30 min for H}_2\text{S-free feed}$	$\mu\text{moles of H}_2\text{S adsorbed in 30 min (averaged capacities)}$
SRE-1	La/Zr=0.8	22	30	120
SRE-5	Tb/Ce/La/Al= 0.2/0.9/1/28	20	42	100
Reom_14	Ce/La=0.3	40	67	51
Reom4_Mn	M/(Ce+La)= 0.1	44	45	160
Reom4_Mn2	M/(Ce+La)= 0.3	57	100	150

**Table 3. 3 Tar removal of naphthalene and sulfur capacities of fresh sorbents**

Sorbent	$\mu\text{moles}$ of tar removed in 30 min with feed containing $\text{H}_2\text{S}$	$\mu\text{moles}$ of $\text{H}_2\text{S}$ adsorbed in 30 min
SRE1	32	120
Reom4_Mn2	24	150

The sulfur capacities and percentages of naphthalene removed are compared in Fig. 3.9. The numbers for  $\mu\text{mol}$  naphthalene removed refer to sorbents already used in multiple sulfidation/TPD cycles and for feeds containing  $\text{H}_2\text{S}$ , as in Table 3.2. The sulfur capacities shown are the average capacities over multiple cycles.



**Figure 3. 9 Comparison of naphthalene removal and sulphur capacities of sorbents**

Among the sorbents tested, the REOs impregnated with Mn have the highest capacity for the simultaneous removal of tars and desulfurization. The capacity for tar removal and desulfurization increased with an increase in the Mn/REO molar ratio from 0.1 to 0.3. The

supported REOs, which showed higher sulphur removal capacities compared to unsupported REOs, showed a lower capacity for tar removal. Among the supported REOs, the one supported on  $ZrO_2$  showed slightly higher tar removal and desulfurization capability compared to those supported on  $Al_2O_3$ . The advantage is even greater if compared on a surface area basis. This may be because lanthana is completely miscible in zirconia and formed a homogeneous solid solution as suggested by XRD. These results suggest that the intimate contact between zirconia and lanthana is favourable for the naphthalene conversion. When the sorbents were tested only for tar removal, without  $H_2S$  in the reaction mixture, all sorbents showed higher tar removal capability as expected. The total tar removed increased significantly, in some cases almost doubling, for all the sorbents except for Reom4\_Mn. Among the fresh sorbents tested, a supported REO such as SRE1 showed slightly higher tar removal than one already used in multiple sulfidation cycles, as might be expected. However, fresh Reom4\_Mn2 actually removed less naphthalene when compared to Reom4\_Mn2 used in multiple sulfidation cycles. This may be because there is some miscibility or spreading of Mn taking place that enhances both the desulfurization capacity and the tar removal capability. Comparing all the sorbents, it can be concluded that Reom4\_Mn2 has the highest capability for simultaneous tar removal and desulfurization of gasifier effluents. This may be because both  $MnO_2$  and lanthana formed a solid solution with ceria, as suggested by the XRD results of REOM\_4 and REOM4\_Mn. But a separate  $MnO_2$  phase was observed in REOM4\_Mn2 (more active) which was absent in REOM4\_Mn (less active). Mn ions are initially incorporated into  $CeO_2$  defect sites, possibly catalyzing the sintering of ceria which took place. Above a critical concentration, Mn then occupies the lattice sites at outer layers of the crystallites (Murugan et al., 2005). Both the ceria-lanthana solid solution and the other unknown form of Mn must be effective in removing tar and

H<sub>2</sub>S, as REOM4\_Mn2 showed better tar and H<sub>2</sub>S removal capability compared to REOM4\_Mn, while simple supported (on Al<sub>2</sub>O<sub>3</sub>) MnO<sub>2</sub> is almost inactive.

GC-MS analyses of the gas phase were performed on REOM4\_Mn with the same gas feed but with N<sub>2</sub> substituting for H<sub>2</sub>S. Samples were collected into gas bags every 10 minutes and injected into a GC-MS. The easily identifiable components in the samples were N<sub>2</sub>, CO<sub>2</sub> and naphthalene. Since all the light gases (N<sub>2</sub>, CO<sub>2</sub> etc.) eluted together on the GC column, it was not possible to distinguish CO, H<sub>2</sub>, ethylene, or propane in the presence of so much N<sub>2</sub> and CO<sub>2</sub>. There were traces of CH<sub>4</sub> and C<sub>2</sub>H<sub>6</sub> in all samples, so C<sub>2</sub>H<sub>4</sub> was probably there in trace amounts also. There was no propylene, so propane was probably not present (within detection limits, which were about 10 ppm). Traces of benzene were also found in all samples.

The initial tar conversion was 10-45% for all the sorbents tested with H<sub>2</sub>S in the feed but the conversion decreased with time. The initial tar conversion was 18-50% for sorbents tested without H<sub>2</sub>S in the feed. All the catalysts tested deactivated in 30 minutes or less. In the case of feeds without H<sub>2</sub>S, this may be due to the formation of coke on the surface of the sorbents blocking the active sites. Complete naphthalene conversion was achieved using a MgO - supported nickel catalyst tested at 1073K using a feed containing 49.8% N<sub>2</sub>, 12% CO, 10% H<sub>2</sub>, 11% CO<sub>2</sub>, 5% CH<sub>4</sub> and 12% H<sub>2</sub>O, 0.3% naphthalene and 100 ppmv H<sub>2</sub>S, with a GHSV of 2080 h<sup>-1</sup>. The catalyst was stable for 100 h (Nacken et al., 2007). The high reforming activity and stability of this catalyst compared to the sorbents tested in this work may be due to the high temperature and low GHSV used, and the small amount of sulfur present.

The tar conversion using Ni/MgO catalysts at 873 K and a feed composition of C<sub>10</sub>H<sub>8</sub>/C<sub>6</sub>H<sub>6</sub>/H<sub>2</sub>O/N<sub>2</sub>/Ar = 0.3/2.7/19.2/10.0/67.8 mol% with a GHSV of 19,200 h<sup>-1</sup> was ~40% for 2 h and decreased to 10% after 10 h reaction (Furusawa et al., 2009). The higher activity and

stability of these catalysts compared to sorbents tested in this work may be due to the absence of both H<sub>2</sub>S and CO<sub>2</sub> in the feed. In the literature most of the tar reforming experiments were done without H<sub>2</sub>S in the feed. Adding H<sub>2</sub>S to the feed should reduce the tar reforming capability of the sorbents since H<sub>2</sub>S also competes for active sites and is adsorbed irreversibly. However, there is no doubt that adding Ni to the present materials would greatly increase the rates of tar cracking or reforming. For example, 90% tar conversion at 1073-1123K was reported for a Ni-WO<sub>3</sub>/MgO-CaO catalyst with a feed consisting of 20% H<sub>2</sub>, 5% CO, 5% CO<sub>2</sub>, 3.5% tar (naphthalene / toluene), 0-500 ppm H<sub>2</sub>S, 18% H<sub>2</sub>O and balance N<sub>2</sub>, with a GHSV of 14000 h<sup>-1</sup>. This catalyst was stable for 100 h (Sato and Shinoda 2007) in the presence of 500 ppm of H<sub>2</sub>S.

The sulfur adsorption capability of the sorbents tested in this work was found to be higher than that of the commercial sorbent CDX. The tar removal capacities of the sorbents tested were lower than some of the catalysts tested in literature. However, adding Ni to the present materials may greatly increase the rates of tar cracking or reforming.



## CHAPTER 4

### CONCLUSIONS

Ce/La oxides, Ce/La/M (M = transition metal) oxides and Ce/La/REO/Al<sub>2</sub>O<sub>3</sub> (REO = a third rare earth oxide) sorbents were studied for the simultaneous desulfurization and tar reforming of synthetic biomass gasifier effluents. These sorbents were prepared by sol-gel and impregnation methods. Surface areas of the sorbents were determined by the BET method and the crystalline phases were determined using XRD. Multiple cycles of sulfidation and regeneration were carried out to evaluate the sulfur removal capacity and stability of the sorbents. Sulfidation tests were done using a simulated gas feed at 903 K. The alumina-supported sorbents retained most of their surface area even when exposed to high temperatures (1103 K) and hence are stable at the high operating temperatures of the gasifiers. Sulfur removal capacity of the pure Ce/La oxide sorbents was relatively low, but in agreement with the literature. Supporting Ce/La oxide on Al<sub>2</sub>O<sub>3</sub> or La oxide on ZrO<sub>2</sub> improved the sulfur removal capacity of the sorbents compared to unsupported REOs, especially when compared on an active (REO) weight basis. Addition of a third rare earth oxide to SREs increased the sulfur removal capacity and the capacity retention of the sorbents. Among all sorbents tested, the Ce/La REOs impregnated with Mn showed the highest sulfur capacities. TPD tests were carried out from 903K to 1103K using He as carrier gas. It can be concluded from the adsorption and desorption capacities of the sorbents that inert gas itself is not sufficient to remove most of the adsorbed sulfur.

Sorbents with higher desulfurization capacities were tested for tar reforming using naphthalene as a simulated tar. The sorbents showed 10-50% tar removal initially but the tar removal decreased as time progressed. All the catalysts deactivated over the same time scale.

Among all the sorbents tested, Reom4\_Mn2 was found to be the best sorbent for simultaneous desulfurization and tar reforming of gasifier effluents.

#### **4.1 Recommendations**

- Regeneration of the sorbents should be carried out with other gas mixtures since regeneration using air is a highly exothermic process which may lead to thermal sintering.
- Characterization of the sorbents after sulfidation and after regeneration must be carried out in order to understand the changes in their surface structure.
- The sorbents must be tested for tar reforming in multiple cycles in order to understand the performance of regenerated catalyst.
- GC-MS tests should be conducted in order to find the products of tar cracking.

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## APPENDIX A - GAS CHROMATOGRAPHY DETAILS

### A.1 Naphthalene cracking

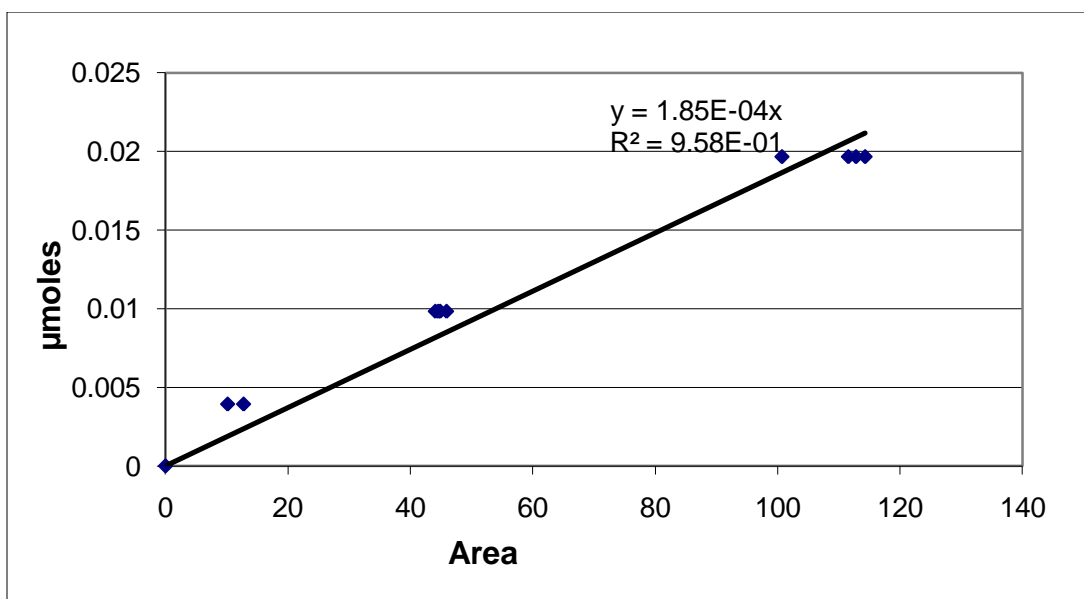
**Table A.1. 1 GC settings for naphthalene cracking analysis**

Parameter	Setting
Column Oven temperature	423K
Detector (FID) temperature	433K
Injection port temperature	473K
Sampling valve temperature	403K
Column	J&W DB-5, 1.5 micron film, 30 m x 0.53mm
Carrier gas	He, 12 mL/min
Auxiliary ga	He, 15 mL/min
Hydrogen flow	35 mL/min
Air flow	340 mL/min
Split flow	He, 20 mL/min
Septum purge flow	He, 2.5 mL/min
Range	2
Attenuation	0

The FID (flame ionization detector) for the HP 5890 Series II GC was calibrated for naphthalene by injecting naphthalene dissolved in heptane solution. 0.02 mol/L of Naphthalene in Heptane solution was prepared. Either 0.2, 0.5, or 1  $\mu$ L of this solution were injected and the samples were analysed as in Table A.1.2 with results in Figure A.1.

**Table A.1. 2 Temperature program for manual injections**

Initial time	1 min
Initial temperature	373 K
Final temperature	423 K
Ramp	5 K/min
Final time	1 min



**Figure A. 1 Naphthalene calibration**

The response factor for naphthalene was calculated as:

$$RF_{\text{naph}} = X_{\text{naph}} / \text{Area}_{\text{naph}}$$

Where  $X_{\text{naph}}$  is micromoles of naphthalene. The response factor was  $1.85 \times 10^{-4} \mu\text{mol/area unit}$  with an  $R^2$  of 0.958.

## A.2. Sulfur compound analysis

For this analysis, a Varian 3800 GC was used. It was equipped with a pulsed flame photometric detector (PFPD) which is specific for sulfur compounds down to sulfur concentrations of ~1 ppmv.

**Table A.2. 1 GC settings for product analysis**

Parameter	Setting
Column Oven temperature	373K
Sampling valve temperature	373K
PFPD detector temperature	473K
TCD detector temperature	473K

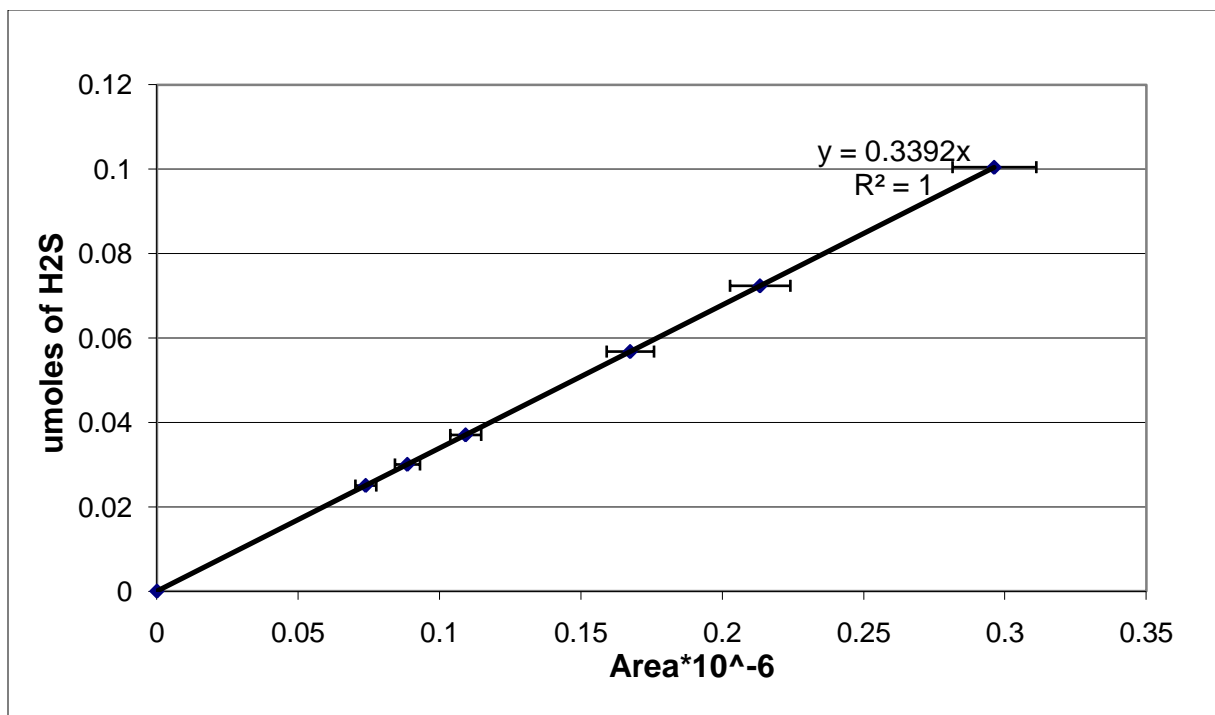
**Table A.2. 2 Varian 3800 settings for sulphur compound analysis**

Column	0.53 mm capillary, no stationary phase
Carrier gas	He, 15mL/min
Air 1	34 mL/min
Air 2	10 mL/min
Hydrogen	20 mL/min
PFPD Range	9.0
PFPD Photomultiplier voltage	630 V
PFPD Gate width	20 ms
PFPD Gate delay	6.0 ms
PFPD Trigger level	220 mV
Sample loop size	2 mL

The PFPD was calibrated using 1% H<sub>2</sub>S/N<sub>2</sub> and CO<sub>2</sub> mixtures; three samples were averaged for each mixture. The response factor (RF) for PFPD was calculated :

$$RF = X_{H_2S} / A_{H_2S}$$

Where X<sub>H<sub>2</sub>S</sub> is micromoles of H<sub>2</sub>S. The response factor was 0.339 μmole/area unit and the results are in Figure A.2.



**Figure A. 2 Calibration for H<sub>2</sub>S**

## VITA

Sumana Adusumilli was born in December 1985, to Jyothi and Lakshmana Rao Adusumilli, in Tenali, Andhra Pradesh, India. She completed her college studies in Vijayawada in 2003. She then joined the Department of Chemical Engineering at Andhra University and earned a Bachelor of Technology in 2007. In August 2007, she was admitted to Louisiana State University, Chemical Engineering Department. There she worked under Dr. Kerry M. Dooley and completed her research work successfully in December 2009. This thesis completes her requirements to receive the degree of Master of Science in Chemical Engineering.